

# High-Temperature Thermal Stability of a Graphene Hall Effect Sensor on Defect-Engineered 4H-SiC(0001)

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**Abstract**—In this letter, we demonstrate a Hall effect sensor in the technology of amorphous- $\text{Al}_2\text{O}_3$ -passivated transfer-free p-type hydrogen-intercalated quasi-free-standing epitaxial Chemical Vapor Deposition graphene on semi-insulating high-purity on-axis 4H-SiC(0001), pre-epitaxially modified with 5-keV hydrogen ( $\text{H}^+$ ) ions. The sensor operates between 305 K and 770 K, with a current-mode sensitivity of  $\sim 75$  V/AT and thermal stability below 0.15 %/K ( $\leq 0.03$  %/K in a narrower range between 305 K and 700 K). It is a promising two-dimensional platform for high-temperature magnetic diagnostics and plasma control systems for modern tokamak fusion reactors.

**Index Terms**—Graphene, SiC, hall effect sensor.

## I. INTRODUCTION

MAGNETIC field sensors are gaining importance as novel applications emerge, including high-performance platforms [1]. Constant progress in nuclear fusion technologies dictates a road map for high-temperature (HT) magnetic diagnostics [2], [3] and sensors operating stably in a wide temperature range, from medium-high (473 K to 623 K) [4], [5], [6], up to the extreme (above 770 K) [7]. Changes in the toroidal magnetic field in industry-driven DEMO-class reactors are about to be monitored with Hall effect sensors

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operating at 473 K (ex-vessel position) and up to 773 K (in-vessel position). The current-mode sensitivity of HT devices strictly depends on the sensor active layer material. Below 673 K, AlGaN/GaN and InAlN/GaN heterostructures offer  $< 100$  V/AT [6], [7], [8], compound-semiconductor thin films  $< 10$  V/AT [4], [5], [9], metal and semimetal Hall sensors (Bi, Sb, Cr, Au, Mo, Ta, Cu)  $< 1$  V/AT [10], [11]. The thermal stability of HT sensors is usually at the level of 0.1 %/K [4], [8], [9], [12].

Earlier, in [12], we reported on the application of hydrogen-intercalated quasi-free-standing (QFS) epitaxial Chemical Vapor Deposition (CVD) graphene on semi-insulating (SI) high-purity (HP) 4H-SiC(0001) in a Hall effect sensor verified between 300 K and 770 K. We reasoned that its performance followed the physics of a double-carrier system governed by holes in the QFS graphene and thermally-activated electrons in the SiC substrate. To restrict the evolution of the electron channel, we introduced the technology of pre-epitaxial modification of the 4H-SiC(0001) surface with hydrogen ( $\text{H}^+$ ) (20 keV and 40 keV) and helium ( $\text{He}^+$ ) (25 keV and 50 keV) ions to intentionally modify its post-epitaxial defect structure. Applying High-Resolution Photo-Induced Transient Spectroscopy, we identified 17 defect levels to prove that the protocol eliminates deep electron traps related to silicon vacancies in the charge state (2-/-) occupying the  $h$  and  $k$  sites of the 4H-SiC lattice, previously recognized as dominant in the process of thermal activation [13].

Of the two ions, the  $\text{H}^+$  served the sensory platform better. Our final recommendation was to reduce the implantation energy to a sub-20-keV level in the hope that the sensitivity and hole mobility curves meet the optimum balance between the magnitude, linearity, and reproducibility. We reasoned that lowering the implantation energy would benefit the mobility curves, but only to the point that the pre-epitaxial modification brought an advantage.

In this communication, we demonstrate thermal stability and uniformity of hole concentration and mobility, sheet resistance, and sensitivity of an optimized Hall effect sensor in the technology of amorphous- $\text{Al}_2\text{O}_3$ -passivated QFS epitaxial CVD graphene on SI HP 4H-SiC(0001), pre-epitaxially bombarded with 5-keV hydrogen ( $\text{H}^+$ ) ions.

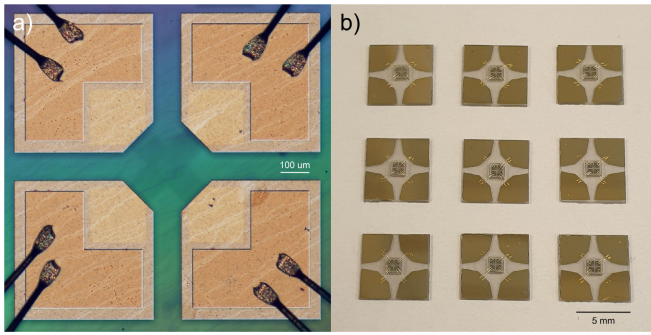


Fig. 1. (a) Top-view optical image of the individual sensor in the defect-engineered  $a\text{-Al}_2\text{O}_3/\text{QFS-graphene}/4\text{H-SiC}(0001)$  technology. (b) Photograph of mounted and wire-bonded sensors.

## II. FABRICATION AND METHODS

### A. Graphene Epitaxy on Defect-Engineered 4H-SiC(0001)

The Hall effect sensor was produced in the graphene-on-SiC technology (GET<sup>®</sup> [14]). The graphene was transfer-free, p-type, *in-situ* hydrogen-intercalated [15], QFS, epitaxial CVD [16], [17], and statistically  $\sim 1.5$ -layer, within the understanding of the relative intensity of the SiC-related Raman-active  $A_1$  mode at  $964\text{ cm}^{-1}$  [18], [19]. It was grown at 1873 K in an Aixtron VP508 reactor on a  $20\text{-mm} \times 20\text{-mm}$  sample diced from a 4-in,  $500\text{-}\mu\text{m}$ -thick, semi-insulating, high-purity, nominally on-axis 4H-SiC(0001) wafer (Wolfspeed Inc.), using thermally decomposed propane as the carbon-rich gas [20]. Before the epitaxy of graphene, the as-purchased 4H-SiC(0001) surface was implanted with a fixed dose of  $1 \times 10^{14}\text{ cm}^{-2}$  of hydrogen ( $\text{H}^+$ ) ions, with an energy of 5 keV, using an apparatus developed by Ionics SA. The expected penetration depth assessed with the Stopping and Range of Ions in Matter program (SRIM) was  $d = 55\text{ nm}$ .

### B. Hall Effect Sensor Technology

The sample was processed into a batch of 96 *van der Pauw* structures [18]. Each structure was a  $1.4\text{-mm} \times 1.4\text{-mm}$  four-terminal device [12] featuring an oxygen-plasma-etched, equal-arm, cross-shaped [21]  $100\text{-}\mu\text{m} \times 300\text{-}\mu\text{m}$  graphene mesa, Ti/Au (10 nm/110 nm) current feed and voltage read-out contacts, and a 100-nm-thick, atomic-layer-deposited, amorphous, non-stoichiometric [22]  $\text{Al}_2\text{O}_3$  passivation [23] synthesized from trimethylaluminum and deionized water at 770 K in the Picosun R-200 Advanced reactor. The choice of this specific geometry, rather than of an optimized Hall bar, was justified by the authors' will to elucidate the transport properties of the 5-keV  $\text{H}^+$ -modified  $a\text{-Al}_2\text{O}_3/\text{QFS-graphene}/4\text{H-SiC}(0001)$  platform. The sample underwent dicing, and nine individual sensors were mounted onto and gold-wire-bonded to in-house-made  $6.6\text{-mm} \times 6.6\text{-mm}$   $364\text{-}\mu\text{m}$ -thick sapphire holders equipped with four Ti/Au (10 nm/190 nm) corner contacts. Fig. 1 (a) illustrates the optical photograph of an individual sensor, while Fig. 1 (b) mounted sensors.

### C. Broad-Temperature Electrical Characterization

The sensors underwent verification under  $I = 1\text{-mA}$  direct current bias in an 0.556-T Ecopia AHT55T5 Hall effect

measurement system in a broad temperature range (305 K to 770 K, with a step of 15 K) and exposure to ambient atmosphere. Their basic electrical properties, including hole concentration  $p_S$ , hole mobility  $\mu_p$ , sheet resistance  $R_S$ , and current-mode sensitivity  $S_I$  (Hall coefficient), were measured and plotted as a function of temperature. The current-mode sensitivity followed Eq. 1:

$$S_I = \frac{dU_{Hall}}{dB} / I = 1 / (p_S e) [V/AT] \quad (1)$$

where  $U_{Hall}$  is Hall voltage,  $B$  is magnetic field,  $I$  is feed current,  $p_S$  is hole concentration, and  $e$  is the unit charge.

To graphically visualize the thermal stability of the parameters, each measurement point was color-coded for its local change with temperature, according to Eq. 2:

$$\alpha_T = \left| \frac{\Delta P}{\Delta T} / P \right| \times 100\% [\%/K] \quad (2)$$

where  $P$  stands for either  $p_S$ ,  $\mu_p$ ,  $R_S$ , or  $S_I$ , and  $T$  is the temperature. An absolute value of the normalized change was chosen, and the scale of  $\alpha_T$  was equalized for all the transport parameters so that areas of corresponding stability could be easily detected.

## III. RESULTS AND DISCUSSION

The thermal stability scale was set between 0.0 %/K and 0.2 %/K for all the measured parameters. Its sign was purposefully ignored as the parameters may have opposite thermal tendencies.

Room-temperature (RT) verification (averaged over the  $50000\text{-}\mu\text{m}^2$  area of the cross-shaped QFS graphene mesa) revealed that the sensors exhibit p-type conductance with the hole density  $p_S$  at the level determined by the double polarization mechanism [22] of the SiC-related positive polarization [24], quantified by vector  $P_0^{4H} = -2.0 \times 10^{-2}\text{ C/m}^2$  [25], and the negative effect of the  $a\text{-Al}_2\text{O}_3$  passivation [22], fixing the  $p_S$  at approximately  $+8 \times 10^{12}\text{ cm}^{-2}$ . The  $p_S$  remains in agreement with the one measured in analogous materials and device technology but on non-modified 4H-SiC(0001) [12].

RT hole density was verified at  $\sim 8.0\text{-}8.7 \times 10^{12}\text{ cm}^{-2}$  ( $p_{Smean} = 8.4 \times 10^{12}\text{ cm}^{-2}$ , standard deviation  $\sigma_p = 0.2 \times 10^{12}\text{ cm}^{-2}$ ). The  $p_S$  has a very stable character throughout the temperature range, with the  $\alpha_T$  not exceeding 0.03 %/K at least up to  $\sim 700\text{ K}$ . Within the 700 - 770 K sub-range,  $\alpha_T$  rises but does not exceed 0.15 %/K (Fig. 2).

RT hole mobility was confirmed at 1800 - 2000  $\text{cm}^2/\text{Vs}$  ( $\mu_{pmean} = 1860\text{ cm}^2/\text{Vs}$ ,  $\sigma_\mu = 104\text{ cm}^2/\text{Vs}$ ). Its thermal stability throughout the temperature range is dynamic. Four conventional sub-ranges are detectable:  $\sim 300\text{ K}$  to  $\sim 400\text{ K}$  ( $\alpha_T \leq 0.11\text{ %/K}$ ),  $\sim 400\text{ K}$  to  $\sim 600\text{ K}$  ( $\alpha_T$  rises to 0.15 %/K),  $\sim 600\text{ K}$  to  $\sim 700\text{ K}$  ( $\alpha_T$  reaches 0.2 %/K), and  $\sim 700\text{ K}$  to  $\sim 770\text{ K}$  ( $\alpha_T$  drops to 0.15 %/K) (Fig. 3).

RT sheet resistance was measured at 350 - 450  $\Omega/\text{sq}$  ( $R_{Smean} = 424\text{ }\Omega/\text{sq}$ ,  $\sigma_R = 56\text{ }\Omega/\text{sq}$ ). The thermal stability of  $R_S$  is equally as dynamic as the stability of  $\mu_p$ , with four detectable sub-ranges, in qualitative and quantitative agreement with the previous observation for  $\mu_p$  (Fig. 4).

RT current-mode sensitivity was confirmed at 72 - 78 V/AT ( $S_{Imean} = 74.4\text{ V/AT}$ ,  $\sigma_S = 2.1\text{ V/AT}$ ). It follows the stable

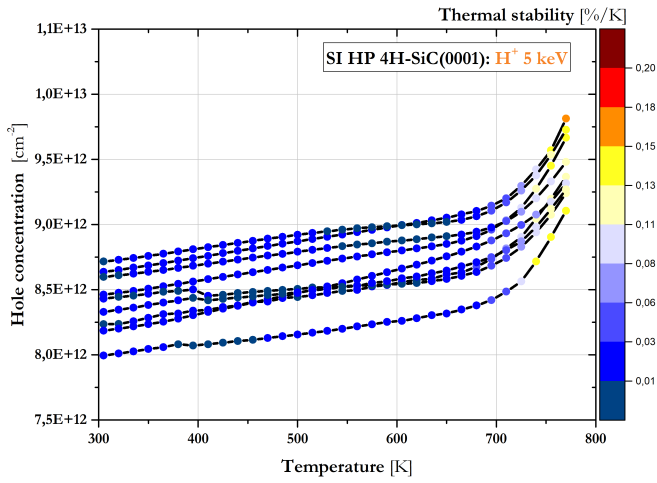


Fig. 2. Hole density profiles as a function of temperature (305 K to 770 K), measured in the nine Hall effect sensors presented in Fig. 1.

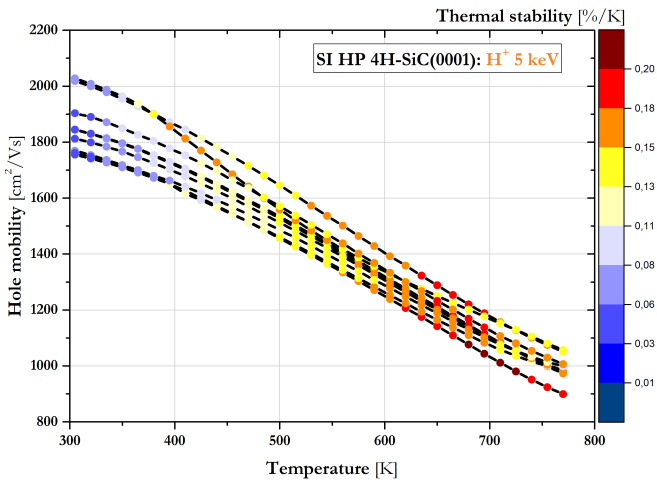


Fig. 3. Hole mobility profiles as a function of temperature (305 K to 770 K), measured in the nine Hall effect sensors presented in Fig. 1.

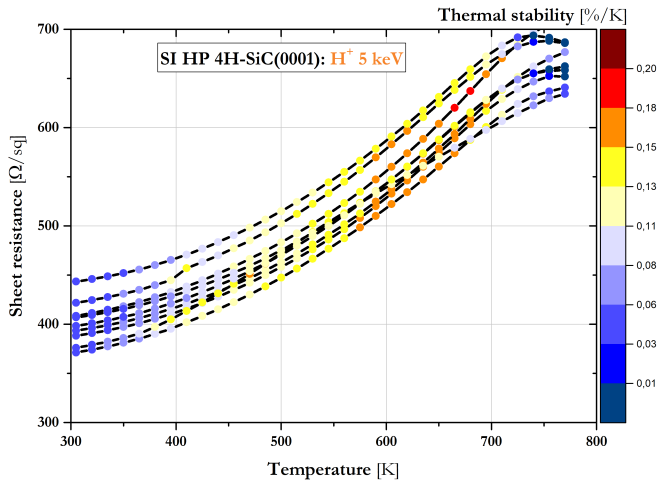


Fig. 4. Sheet resistance profiles as a function of temperature (305 K to 770 K), measured in the nine Hall effect sensors presented in Fig. 1.

character of  $p_S$ , with the  $\alpha_T$  not exceeding 0.03 %/K at least up to  $\sim 700$  K ( $S_{I\text{mean}} = 70.6$  V/AT,  $\sigma_S = 1.8$  V/AT) and rising but not exceeding 0.15 %/K in the 700 - 770 K ( $S_{I\text{mean}} = 66.1$  V/AT,  $\sigma = 1.7$  V/AT) sub-range (Fig. 5). It is apparent that the undesirable contribution of the SiC-related electron

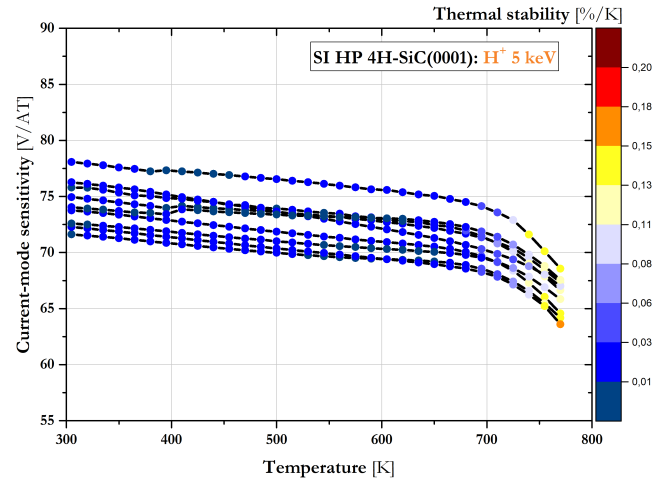


Fig. 5. Current-mode sensitivity as a function of temperature (305 K to 770 K), measured in the nine Hall effect sensors presented in Fig. 1.

channel to the double-carrier conductivity, is suppressed up to  $\sim 700$  K.

At 305 K, the thermal profiles display a percentile spread around their mean value ( $\sigma/\text{mean} \times 100\%$ ) of  $p_S$  (2.8 %),  $\mu_p$  (5.6 %),  $R_S$  (13.2 %), and  $S_I$  (2.8 %). At 770 K, the dispersion is down to  $p_S$  (2.6 %),  $\mu_p$  (4.7 %),  $R_S$  (10.4 %), and  $S_I$  (2.6 %). This low variation is possible only because direct epitaxial growth does not require a polymer-assisted transfer [26]; it is elementally pure and void of substitutional dopants, and its charge carrier type and density are determined solely by the SiC-related vector of spontaneous polarization. The vestigial dispersion is traced to the developed topography of epitaxial graphene, marked with a fingerprint-like combination of micrometer-scale terraces and nanometer-scale steps [19], [27]. It remains an inherent feature of this technology and the primary source of offset voltage (here, at 305 K: from  $\sim 6$  mV to  $\sim 83$  mV, with one sensor exhibiting  $\sim 224$  mV, and at 770 K: from  $\sim 10$  mV to  $\sim 121$  mV, with the one sensor proving  $\sim 366$  mV).

#### IV. CONCLUSION

The Hall effect sensor fabricated in the technology of amorphous- $\text{Al}_2\text{O}_3$ -passivated QFS epitaxial CVD graphene on defect-engineered semi-insulating high-purity nominally on-axis 4H-SiC(0001), pre-epitaxially implanted with 5-keV hydrogen ( $\text{H}^+$ ) ions, operates between 305 K and 770 K and offers current-mode sensitivity of  $\sim 75$  V/AT. Throughout this temperature range, its hole mobility and sheet resistance maintain thermal stability  $\alpha_T \leq 0.2$  %/K, while its hole density and current-mode sensitivity  $\alpha_T \leq 0.15$  %/K. In a narrower range, between 305 K and 700 K, the stability of the current-mode sensitivity  $\alpha_T \leq 0.03$  %/K. The sensor is a promising two-dimensional platform for high-temperature magnetic diagnostics and plasma control systems for modern tokamak fusion reactors.

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