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RESEARCH ARTICLE

Improvement of Retention Characteristics Using Doped SiN Layer Between WL Spaces in 3D NAND Flash

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ABSTRACT We propose a novel structure of a charge trapping layer, that is doped between Word Line(WL) spaces in 3D NAND flash memory. To estimate the retention characteristics, the ΔV_{th} of each structure by doping type is compared to a reference structure during retention operation. In this study, lateral charge spreading, rather than vertical loss, is mainly studied as an indicator to determine how long stored data can survive on the selected cell. When the SiN layer is doped with p-type, the electrostatic potential decreases and SRH recombination increases in the doping region. As a result, the p-type doping structure has better retention characteristics than others. In addition, for an optimized structure between retention characteristics and cell current, the doping region of the SiN layer is split and analyzed by the thickness of the doping region. Since the p-type doping region is thicker, lateral spreading is effectively blocked although the cell current decreased slightly.

INDEX TERMS 3D NAND, doped SiN layer, electrostatic potential, lateral charge spreading, SRH recombination.

I. INTRODUCTION

The advancement of NAND flash memory from a 2D planar structure to a 3D vertical structure enables faster speed at a lower cost, higher density, and scaling down. Although 3D NAND charge trap flash(CTF) has many advantages and applications [1], [2], [3], [4], [5], there is a critical problem caused by sharing the charge trap flash (CTL) of the target cell with adjacent cells [6], [7], [8], [9], [10], [11]. In CTL, there is charge loss during retention operations, due to vertical loss and lateral charge spreading. The cause of vertical loss is the trapped charges of nitride passing through the tunneling oxide [12], [13], [14]. Lateral charge spreading is caused by the electric field [15], [16] from charges in WL spaces and adjacent cells. Charge loss has a greater impact on 3D NAND flash memory as it is scaled down, and there is a steady stream of research about this [17], [18], [19], [20], [21], [22]. In this

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work, we propose a new structure of 3D NAND CTF with thin doping in the SiN layer (CTL) between WL spaces using high SiN/SiO₂ etching selectivity [23]. These doped regions in the SiN layer reduce the impact of the electric field causing lateral charge spreading. The study consists of two parts, a comparison of characteristics by doping type and by doping thickness. In the first part about doping type, we show which doping type in SiN improves the retention characteristics. Then, in the second part about doping thickness, the optimal doping thickness is found using the relationship between the retention characteristics and the cell current.

II. PROPOSED STRUCTURE

Fig. 1 (a) shows a cross-sectional view of 3D NAND flash memory created by TCAD Sentaurus Device Editor(SDE) [24], [25]. It consists of 3 WLs, an ONO stack, a polysilicon channel, a source, a drain, and an oxide filler. The O/N(=Oxide/Nitride) pitch of the word line is 50 nm for 25 nm. The oxide and SiN layer thicknesses of the ONO stack

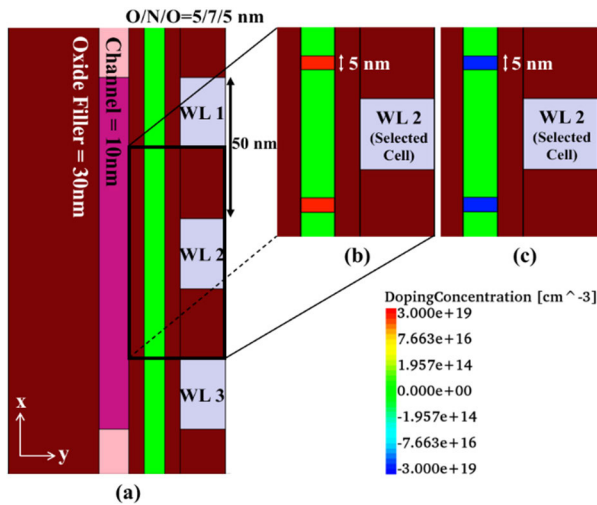


FIGURE 1. (a) Cross-sectional view of reference structure with 3 WLs in 3D NAND flash memory used in TCAD simulation (b) Proposed 3D NAND structure with N-doping in the SiN layer and (c) P-doping in the SiN layer.

are 5 nm and 7 nm, respectively. The source on the bottom of the channel and drain on top of the channel are doped with phosphorus $5 \times 10^{18} \text{ cm}^{-3}$ and a channel is doped with boron $1 \times 10^{15} \text{ cm}^{-3}$. The SiN layer of the ONO stack between WL spaces is doped with phosphorus in Fig. 1 (b) and boron in Fig. 1 (c). The doping concentration is both $3 \times 10^{19} \text{ cm}^{-3}$ which is the same value as the trap concentration of the SiN layer set in the TCAD simulation. The length of doping in SiN is 5 nm in the x-direction, and the thickness of doping in SiN is 7 nm in the y-direction.

Fig. 2 illustrates the manufacturing process of proposed 3D NAND flash structures in Fig. 1 (b) and (c), which incorporate doped SiN regions. In Fig. 2 (b) and (c), a 10 nm oxide layer is uniformly deposited and etched using dry etching techniques [26]. Moving on to Fig. 2 (d), the SiN is ultimately etched by 5 nm in the x-direction. For the deposition and etch process of oxide and SiN, thin and uniform structures between cells can be created with control of elaborate process conditions such as ALD cycle or etch rate [27], [28]. In Fig. 2 (e), the doped SiN layer is deposited using PECVD or ALD [29], [30], [31], [32] with uniform distribution. For this process, diborane or phosphine can be introduced into the chamber as dopants for N and P-doped SiN layer.

Threshold voltage (V_{th}) is measured with a constant current method with 100 nA. The nonlocal tunneling model is used between the channel and tunneling oxide. Also, the Poole-Frenkel effect for trap-assisted electron transport and Shockley-Read-Hall (SRH) recombination model for trap-assisted recombination are applied. The operation conditions for erase, program and read are shown in Table 1.

III. RESULTS AND DISCUSSION

A. TYPES OF DOPING

Fig. 3 shows the results of ΔV_{th} over a retention time at 300 K as a parameter of the doping types in the SiN. The program V_{th} of selected WL cell is set to 4 V and erase V_{th} of

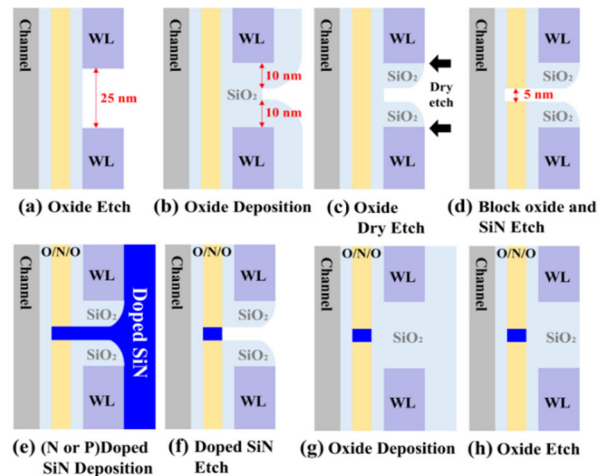


FIGURE 2. Manufacturing process of 3D NAND flash structures with doped SiN layer.

TABLE 1. Conditions for TCAD simulation.

	B/L	Selected cell	Unselected cell	Temperature
<i>Erase</i>	0 V	-18 V	0 V	-
<i>Program</i>	0 V	18 V	7 V	-
<i>Read</i>	0 to 0.5 V	-5 to 10 V	7 V	-
<i>Retention</i>	0 V	0 V	0 V	300 K

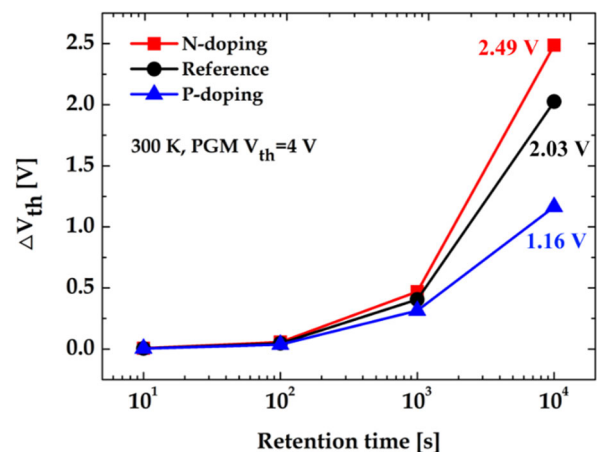


FIGURE 3. Comparison of ΔV_{th} ($= |V_{th_retention} - V_{th_program}|$) as a function of retention time and doping type in the SiN layer.

unselected WL cells is set to -1 V. Three types of structures were introduced including the reference, n-type doping and p-type doping in Fig. 1. The reference of Fig. 1 (a) is the basic structure of 3D NAND flash to compare data with doped structures in Fig. 1 (b) and (c). Fig. 3 shows the retention characteristics from 10 s to 10^4 s. As doping types between WLs space change from n-doping to p-doping, it is confirmed that the retention characteristics are improved as shown in Fig. 3.

This result can be explained by the electrostatic potential in the SiN layer at retention time (10 s) as shown in Fig. 4 (a)

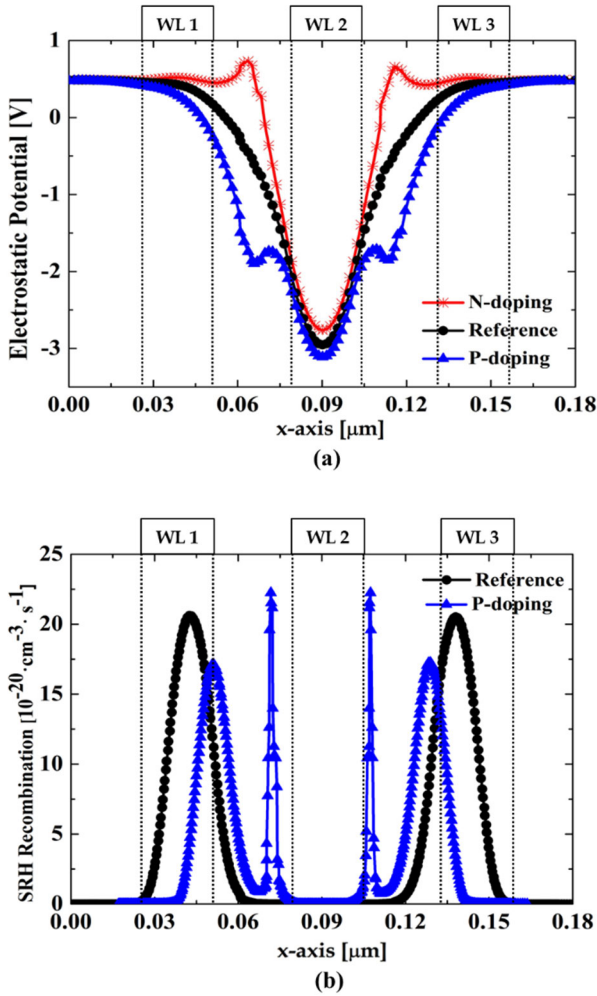


FIGURE 4. (a) Electrostatic potential of SiN layer and (b) SRH recombination of SiN layer according to x-axis for retention time=10 s.

that decreases rapidly in the p-type doping region compared to others during the retention operation. These changes of potential affect lateral spreading. Therefore, electrons trapped for selected cells spread less to adjacent cells due to reduced potential in the p-type doping region. On the contrary, electrons for selected cells spread more to adjacent cells due to increased potential in the n-type doping region. Fig. 4 (b) shows SRH recombination in the SiN layer under the same conditions as shown in Fig. 4 (a). In particular, SRH recombination occurs in unselected cells and p-type doping region. Because many holes exist in the SiN layer in erased WL 1 and WL 3, they contribute to recombination in SiN layer of unselected cells.

Fig. 5 describes the lateral spreading of trapped electron and SRH recombination in the SiN layer during retention operation. In Fig. 5 (a), the trapped electrons in a selected cell(WL 2) move to unselected cells(WL 1 and WL 3) during the retention operation, and they pass through the p-type doping region. As recombination occurs between moving electrons and holes by p-type doping in Fig. 5 (b), the number of electrons spreading to the adjacent cells decreases, which

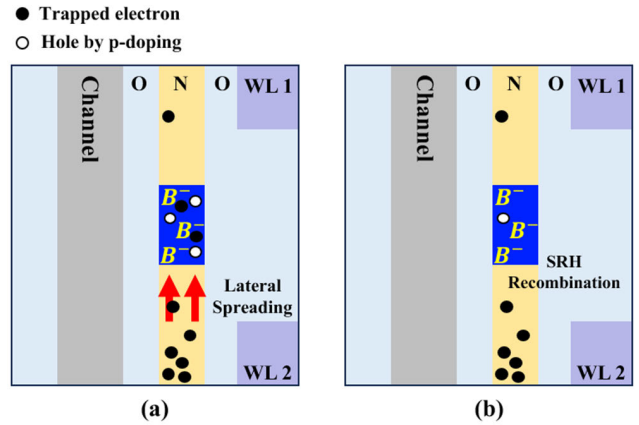


FIGURE 5. (a) Description of lateral spreading during the retention operation in the p-type doping structure 3D NAND flash memory. (b) Description of SRH recombination of SiN layer between spreading electrons and holes by p-type doping.

means that the retention characteristics were improved. As a result, the difference between ΔV_{th} of p-type (=1.16 V) and ΔV_{th} of reference (=2.03 V) in Fig. 3 is 0.87 V after 10^4 s of retention. That is almost twice the difference between ΔV_{th} of n-type(=2.49 V) and ΔV_{th} of reference(=2.03 V), which is 0.46 V. The reason is that the n-type doping structure is only affected by potential, but the p-type doping structure is affected by both decreased potential and increased SRH recombination in doping regions. Therefore, the retention characteristics for the p-type doping structure are effectively improved.

Fig. 6 (a), (b), and (c) show electron trapped charge distribution in the SiN layer after program operation in n-type doping, reference, and p-type doping structures, respectively. Even if p-type doping structure has the least number of trapped electrons, it shows the largest V_{th} by a small difference when program voltage changes from 20 V to 26 V as shown in Fig. 6 (d). This result can be explained with Fig. 5. In Fig. 5 (a), there are the same number of acceptors (boron) and holes in the p-type doping region by ionization [33], [34]. As the lateral spreading of electrons occurs, the number of holes decreases due to recombination as shown in Fig. 5 (b). Therefore, p-type doping region has a negative charge due to more acceptors. This causes higher threshold voltage for the read operation and also affects the cell current as shown in Fig. 7. Even if p-doping structure shows the smallest cell current, the difference between reference and p-doping structure is negligible, which is about 0.2 mA/ μ m.

B. THICKNESS OF DOPING

In the previous study on types of doping, p-type doping structure had the best retention characteristics, but the cell current characteristics were degraded. The thinner a p-type doping region is, the less it is affected by the acceptor which means that it is good to improve cell current. In addition, it is more effective for the doping region to be located on the right side of the SiN layer based on Fig. 8. Fig. 8 shows the E-Field distribution of the reference structure during the retention

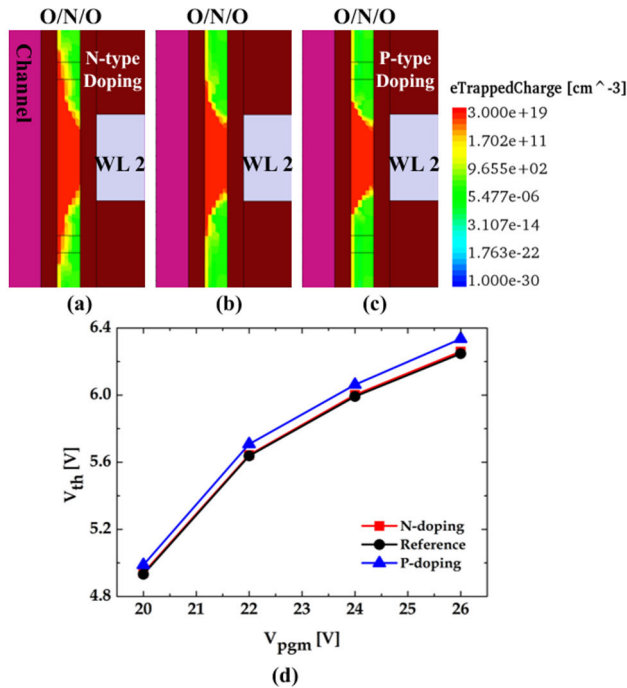


FIGURE 6. Electron trapped charge distribution in the SiN layer after the program of (a) n-doping in SiN, (b) reference, and (c) p-doping in SiN structures. (d) Comparison of V_{th} as a function of doping type in the SiN layer and program voltage from 20 V to 26 V.

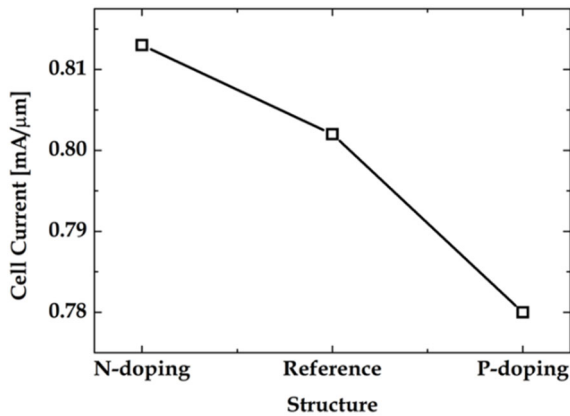


FIGURE 7. Cell current according to doping type in the SiN layer for drain voltage=9 V.

operation. Fig. 9 (a) and (b) show the charge distribution where the retention time is 10 s and 10^4 s, respectively. Fig. 9 (c) shows the electron trapped charge in the SiN layer along the line from point A to B in Fig. 9 (a) and (b). It is observed that lateral spreading occurs more on the right side near point B than the left side near point A in SiN as a stronger electric field is formed on the right side of SiN in Fig. 8.

For the optimized point, the thickness of the p-doping region is split in the y-direction from 1 nm to 7 nm in Fig. 10. The SiN layer is doped from the right side to the left side to effectively block lateral spreading using the minimum thickness of the doping region. Fig. 11 shows the results of ΔV_{th} over retention time at 300 K as a parameter of the doping thickness in the SiN. It is observed that the slope

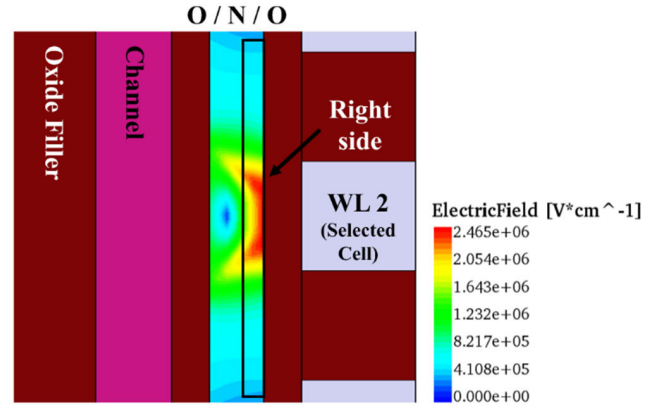


FIGURE 8. Schematic(TCAD) of electric field in the SiN layer during the retention operation at retention time=10 s in the reference structure.

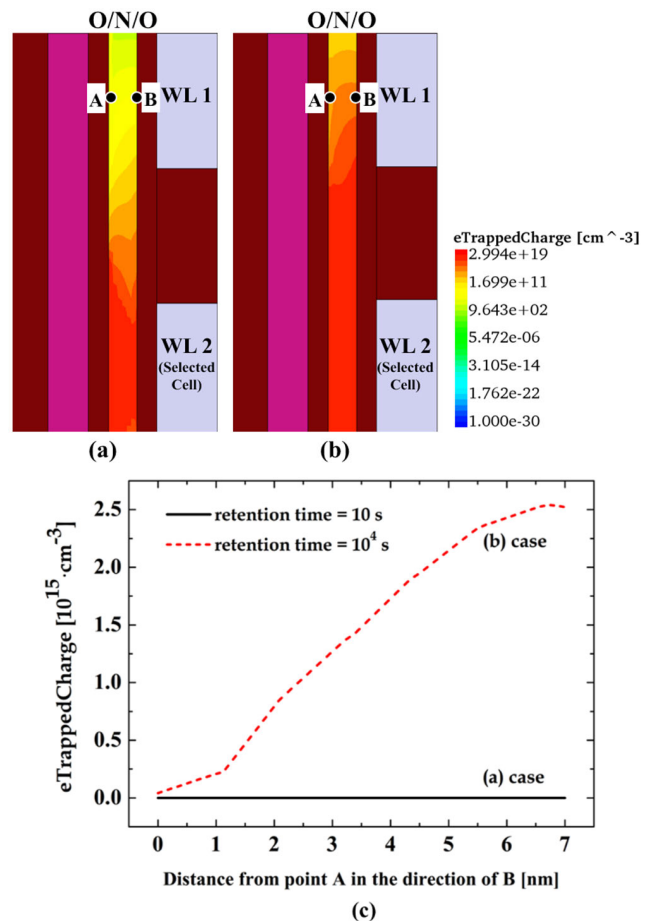


FIGURE 9. (a) Schematic(TCAD) of charge distribution in the reference structure for retention time=10 s and (b) 10000 s. (c) electron trapped charge in SiN as a function of distance from point A in the direction of B of Fig. 9 (a) and (b).

decreases at a slower rate as the doping region becomes thicker. Fig. 12 shows the p-type doping region's thickness window between $1/\Delta V_{th}$ using the data of Fig. 11 and the cell current estimated where the drain voltage is 9 V. The $1/\Delta V_{th}$ increases about 58 % under the condition of 7 nm thickness compared to 1 nm, while the cell current decreases only about 2 %, which is negligible. As a result, the best

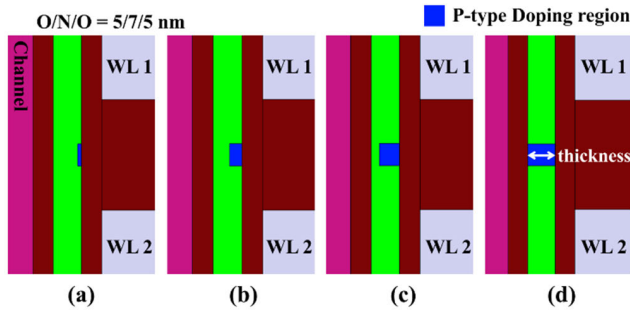


FIGURE 10. Schematic(TCAD) with variations of thickness of the p-type doping region in SiN for (a) 1 nm, (b) 3 nm, (c) 5 nm, and (d) 7 nm.

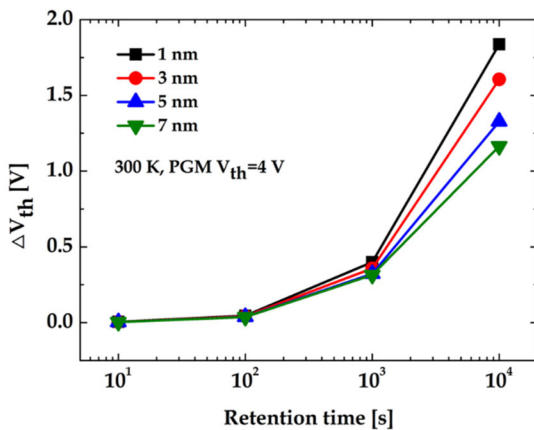


FIGURE 11. Comparison of ΔV_{th} ($= |V_{th_retention} - V_{th_program}$) as a function of retention time and doping thickness in the SiN layer.

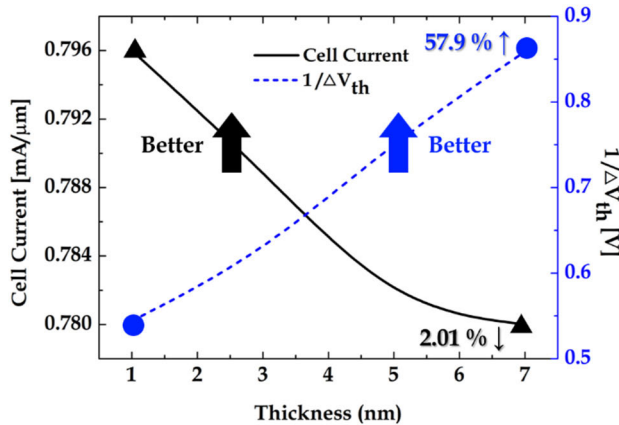


FIGURE 12. Cell current for 9 V of drain voltage and $1/\Delta V_{th}$ with the variation of the thickness of the p-type doping region of SiN layer.

window for thickness of the p-type doping region is between 3 nm and 4 nm considering both the retention characteristics and the cell current.

IV. CONCLUSION

A new structure using the doped SiN layer of 3D NAND flash was proposed. The p-type doping structure has the best retention characteristic of three structures: reference (Fig. 1 (a)), n-doping (Fig. 1 (b)), and p-doping (Fig. 1 (c)), due to decreased potential and increased SRH recombination of holes in the doping region. This effectively reduces lateral

charge spreading from a selected cell to adjacent cells. On the other hand, p-doping structure degrades the cell current due to more acceptors as we discussed earlier. To solve this problem, an additional simulation was performed which shows the trade-off between cell current and retention characteristics. Finally, we found out the optimal thickness of p-doping region, which is between 3 nm to 4 nm, to meet proper cell current and retention characteristics.

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