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Investigations on Wide-Gap Al_{0.21}Ga_{0.79}N Channel MOS-HFETs With In_{0.12}Al_{0.76}Ga_{0.12}N Barrier/Buffer and Drain Field-Plate

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ABSTRACT This work investigates, for the first time, wide-gap Al_{0.21}Ga_{0.79}N channel metal-oxidesemiconductor heterostructure field-effect transistors (MOS-HFETs) with In_{0.12}Al_{0.76}Ga_{0.12}N barrier/buffer and drain field-plate (DFP) designs. High-k and wide-gap Al₂O₃ was grown as the gate oxide and surface passivation by using non-vacuum ultrasonic spray pyrolysis deposition (USPD) technique. A control device having the same epitaxial layers, except with In_{0.12}Al_{0.88}N barrier/buffer was studied in comparison. Enhanced spontaneous polarization effect, improved interfacial quality, and enhanced carrier confinement have been achieved by using the In_{0.12}Al_{0.76}Ga_{0.12}N barrier/buffer design, which has successfully resulted in improved carrier transport, increased electron concentration, and high current densities. The present In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/Al_{0.21}Ga_{0.79}N MOS-HFET design with (without) DFP design has demonstrated superior maximum drain-source current density ($I_{DS,max}$) of >1 (>1) A/mm at $V_{DS} = 20$ V, high saturated drain-source current density at $V_{GS} = 0$ V (I_{DS50}) of 791.1 (755) mA/mm, and low specific on-resistance ($R_{on,sp}$) of 2.83 (2.81) mΩ/cm². High device figure-of-merit (FOM) on $BV_{DS}^2/R_{on,sp}$ of 93.7 (75.4) MW/cm² was also obtained with the three-terminal on-state drain-source breakdown voltage (BV_{DS}) of 515 (460) V. The present design is promisingly advantageous to high-current and high-voltage power-switching circuit applications.

INDEX TERMS Wide-gap AlGaN channel, InAlGaN barrier, MOS-HFET, Al₂O₃, non-vacuum ultrasonic spray pyrolysis deposition, drain field-plate.

I. INTRODUCTION

With the increasing demands of high-voltage power- switching devices for applications of DC-DC converters [1], [2], data centers [3], vehicle electronics [4], and renewable energy [5], heterostructure field-effect transistors (HFETs) with wide-gap channel have been explored [6], [7], [8], [9]. As compared to GaN-based devices, the widegap AlGaN compounds possessing high Johnson's figureof-merit (JFOM) [10] and high Baliga's figure-of-merit (BFOM) [11] are suitable channel recipes to enhance the high-voltage operation. In order to improve battery charging efficiency for electric vehicles [12] and enhance power performance for wireless communication [13], [14] applications, devices with high current drive capability are needed. Our previous works have studied Sidoped AlGaN channel MOS-HFETs [15], [16] showing depletion-mode operation and F⁻-implanted AlGaN barrier MOS-HFET [17] exhibiting enhancement-mode characteristics. This work presents, for the first time, wide-gap Al_{0.21}Ga_{0.79}N channel MOS-HFETs with designs of In_{0.12}Al_{0.76}Ga_{0.12}N barrier/buffer and drain field-plate (DFP). A control In_{0.12}Al_{0.88}N/AlN/ Al_{0.21}Ga_{0.79}N device was fabricated and characterized in comparison. It is known that InAlN/GaN heterostructure possesses lattice match and enhanced spontaneous polarization effects [18], [19] as compared to AlGaN/GaN heterostructure. Yet, the carrier



FIGURE 1. Schematic device structures of (a) the control sample A1 and (b)-(c) the present MOS-HFET designs of samples B1-B2.

transport would be degraded by the heterointerface defects and alloy scattering caused by different growth temperatures between InN and AlN during epitaxy. This work explores a novel In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/ Al_{0.21}Ga_{0.79}N heterostructural design to improve the carrier transport with preserved good polarization effect. Besides, a wide-gap In_{0.12}Al_{0.76}Ga_{0.12}N (In_{0.12}Al_{0.88}N) buffer below the conduction channel was devised in the present (control) devices to enhance the carrier confinement, which would further increase both the electron concentration and current density. In addition, designs of MOS-gate, high-k gate dielectric, oxide passivation, and DFP [20], [21] were integrated to improve the gate modulation capability, reduce gate leakages, alleviate gate-drain electric field, and improve breakdown performance. Al₂O₃ was grown as gate dielectric and surface passivation layer at the same time by using a costeffective non-vacuum ultrasonic spray pyrolysis deposition (USPD) [22], [23] technique.

II. MATERIAL GROWTH AND DEVICE FABRICATION

Figs. 1(a)-(c) show the schematic device structures of (a) the control In_{0.12}Al_{0.88}N/AlN/Al_{0.21}Ga_{0.79}N MOS-HFET (sample A1) and (b)/(c) the present In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/Al_{0.21}Ga_{0.79}N MOS-HFET designs with/without DFP (samples B1/B2), respectively. The epitaxial structures were grown on a SiC substrate by using a low-pressure metal-organic chemical vapor deposition (LP-MOCVD) system. Both samples B1 and B2 have the same layer structures, including the GaN nucleation layer, an intrinsic 10-nm In_{0.12}Al_{0.76}Ga_{0.12}N buffer, an intrinsic 20-nm Al_{0.21}Ga_{0.79}N channel, an intrinsic 1-nm AlN layer, an intrinsic 10-nm In_{0.12}Al_{0.76}Ga_{0.12}N barrier, and a 2-nm Si-doped ($\sim 3 \times 10^{18}$ cm⁻³) GaN capping layer, as shown in Figs. 1(b) and (c). The control sample A1 has the identical epitaxial

structures with respect to the present samples B1-B2, except replacing with the $In_{0.12}Al_{0.88}N$ barrier and buffer, as shown in Fig. 1(a).

All the three devices were fabricated at the same time. Standard photo-lithography and lift-off techniques were used for device fabrication [24]. For sample B2, mesa etching was first performed to provide electrical isolation for neighboring devices by using an inductively coupled-plasma reactive ion etcher (ICP-RIE). The etching gas is BCl₃ with a flow rate of 40 sccm. The ICP/RF power settings are both 110/110 W. The chamber pressure is 1 pa and the etching time is 500 seconds. Dry etching was conducted after photolithography to remove the GaN capper between source and drain electrodes. The flow rates of the BCl₃/Cl₂ mixture gases are 10/5 sccm, with the ICP/RF power settings of 100/12 W. The etching time is 80 seconds under chamber pressure of 1 pa. Metal stacks of Ti (20 nm)/Al (100 nm)/ Ni (20 nm)/Au (70 nm) were then evaporated as the source/drain electrodes. The source/drain ohmic contacts were formed by annealing the sample at 950°C for 60 seconds by using a ULVAC MILA-5000 rapid thermal annealing (RTA) system. Then, a 30-nm Al₂O₃ was grown on the exposed barrier surface between source and drain electrodes by using the USPD technique. Finally, after the photolithography to expose the gate/DFP windows, metal stacks of Ni (150 nm)/Au (40 nm) were evaporated on the Al₂O₃ surface to form both the gate and DFP structures, as shown in Fig. 1(c). The same fabrication procedures were applied to samples A1 and B1, except without the formation of DFP, as shown in Fig. 1(a) and 1(b). Besides, all three samples were gate recessed to the same depth by performing the dry etching simultaneously. The studied devices have the same gate length (L_G) of 2 μ m, gate-to-source spacing of 2 μ m, and gate-to-drain spacing of 10 μ m. For sample B2, the DFP length (L_{DFP}) is 2 μ m, which has resulted in an effective gate-to-drain separation of 8 μ m.

The secondary ion-mass spectroscopy (SIMS) profiles of (a) the epitaxial structure (sample A) for the control device and (b) the present epitaxial design (sample B) for samples B1-B2, as shown in Figs. 2(a)-(b), respectively. Under the n-GaN capping layer, apparent decreases in the Gacomposition distribution were observed in sample A, as compared to sample B. The composition variations of the epitaxial structures were verified with the sample designs. The cross-sectional transmission electron microscopy (TEM) photo of the MOS-gate of sample B2 is also shown in Fig. 2(c). The layer thickness of the USPD-grown Al_2O_3 was verified to be 30 nm. Figs. 3(a) and 3(b) show the atomic force microscopy (AFM) photos of surface morphologies for the In_{0.12}Al_{0.76}Ga_{0.12}N barrier of sample B2 after the gaterecess etching and further after the deposition of Al₂O₃ by using USPD, respectively. The average root-mean-square surface roughness after (before) the USPD-deposited Al₂O₃ was determined to be 0.6 (0.77) nm. The surface flatness was apparently degraded by using ICP-RIE and, then, was effectively improved by using USPD.



FIGURE 2. SIMS profiles of the epitaxial structures of samples (a) A and (b) B; (c) the TEM photo of MOS-gate for the present sample B2.



FIGURE 3. AFM photos of surface morphologies for the $In_{0.12}AI_{0.76}Ga_{0.12}N$ barrier of sample B2 (a) after the gate-recess etching and (b) further after deposition of AI_2O_3 by using USPD.

III. EXPERIMENTAL RESULTS AND DISCUSSION

Hall measurements were conducted for both samples A and B at 300 K under a magnetic field of 5000 G. The electron mobility (μ_n), two-dimensional electron gas concentration (n_{2DEG}), and he μ_n - n_{2DEG} product were found to be 417 (282) cm²/V-sec, 3.6 (3.5) × 10¹³ cm⁻², and 1.5 (0.99) × 10¹⁶ (V-sec)⁻¹ for sample B (A). Samples A-B have similar n_{2DEG} concentrations since the same In-ratio of 0.12 was devised for the In_{0.12}Al_{0.88}N and In_{0.12}Al_{0.76}Ga_{0.12}N barriers. Similar polarization effects were maintained to provide appropriate comparison for the studied devices. Due to enhanced carrier confinement by the devised



FIGURE 4. The measured C-V hysteresis characteristics for samples (a) A and (b) B at 300 K.



FIGURE 5. 1/f noise spectra of samples A1 and B1.

wide-gap In_{0.12}Al_{0.76}Ga_{0.12}N or In_{0.12}Al_{0.88}N barrier/buffer, the obtained n_{2DEG} values are higher than our previous works [15], [16] and other InAlN/AlGaN device [25]. Moreover, the μ_n and μ_n - n_{2DEG} product of sample B have been significantly enhanced due to the improved interfacial property in the In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/Al_{0.21}Ga_{0.79}N heterostructure. High current densities are expected for the present In_{0.12}Al_{0.88}N/AlN/Al_{0.21}Ga_{0.79}N MOS-HFET designs.

The measured C-V hysteresis characteristics for the fabricated MOS diodes for samples A and B are shown in Figs. 4(a)-(b), respectively. The diode areas are the same of 31400 μ m². The applied voltage was increased from -9 V to 6 V and swept back immediately to the starting voltage. The hysteresis voltage (ΔV) was defined to be the bias difference between mid-points of the C-V curves. The ΔV was determined to be about 1 (37) mV for sample B (A). Significant decrease in ΔV for sample B, as compared to sample A, has been obtained, indicating the interface property was effectively improved by using the In_{0.12}Al_{0.76}Ga_{0.12}N barrier. Fig. 5 shows the 1/f noise spectra of samples A1 and B1 biased at $V_{GS} = -7.5$ V and -5.5 V with $V_{DS} = -0.5$ V, measured by using an Agilent 35670A amplifier and a BTA 9812B spectrum analyzer. The Hooge coefficients (α_H) at f = 100 Hz were determined [26] to be 1.4×10^{-5} and 2.3×10^{-6} for samples A1 and B1 with the corresponding noise densities of 1.2×10^{-17} and 2.9×10^{-18} Hz⁻¹. Lower 1/f noise floor and lower α_H of sample B1 than those of sample A1 also indicate its improved interfacial quality. The surface electron trapping phenomenon [27]



FIGURE 6. Common-source I_{DS} - V_{DS} characteristics (left) at 300 K and the transfer g_m and I_{DS} as functions of V_{GS} (right) for samples (a) A1, (b) B1, and (c) B2, respectively.

was effectively reduced by the oxide passivation by using USPD. The comparisons of both ΔV and α_H are consistent with the observed enhanced μ_n behavior in the devised In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/Al_{0.21}Ga_{0.79}N heterostructure. Besides, the transfer length method (TLM) [28] was employed to measure the specific contact resistivity (ρ_c) and contact resistances (R_C). ρ_c and R_C were characterized to be 8.7 × 10⁻⁶ (1.8 × 10⁻⁵) Ω -cm² and 0.18 (0.26) Ω -mm for samples B1-B2 (A1), which are much lower than those wide-gap channel devices of our previous works [15], [16]. It is mainly contributed by the devised n-GaN capping layer to greatly improve the source/drain ohmic contacts. The improved interfacial quality by using the In_{0.12}Al_{0.76}Ga_{0.12}N barrier has further improved the ohmic contact characteristics.

The common-source current-voltage $(I_{DS}-V_{DS})$ characteristics source/drain ohmic contacts. The improved interfacial quality (left) at 300 K and the transfer extrinsic transconductance (g_m) and saturated drain-source current (I_{DS}) density vs. the applied V_{GS} for samples A1, B1, and B2 are shown in Figs. 6(a)-(c), respectively. The applied V_{GS} bias was increased from -8 (-12) V to 6 (6) V at 2 (3) V/step for samples B1-B2 (A1) by using a KEITHLEY 4200 analyzer. All the studied devices have shown good pinch-off phenomena, indicating the good gate modulation capability by the high-k MOS-gate structure. As expected by the Hall data, the enhanced current densities in samples B1-B2 have exceeded the instrumentation limit of the KEITHLEY 4200 analyzer, as observed in Figs. 6(b)-(c). The measured I_{DS} at V_{GS} = 0 V (I_{DSS0}) and maximum I_{DS} ($I_{DS,max}$) densities at V_{DS} = 20 V were found to be 628 mA/mm and 978.5 mA/mm, 755 mA/mm and >1 A/mm, and 791.1 mA/mm and >1 A/mm for samples A1, B1, and B2, respectively. The corresponding maximum extrinsic transconductance $(g_{m,max})$ values are 89.7 mS/mm, 148.3 mS/mm, and 150.8 mS/mm. Both samples B1 and B2 have demonstrated superior current drive capability. It was contributed by the improved channel conductivity by the devised In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/ Al_{0.21}Ga_{0.79}N heterostructure. Sample B2 (B1) has shown about 26% (20%) and 68% (65%) improvements in I_{DSS0} and $g_{m,max}$ performances as compared to sample A1. Higher I_{DSS0} density of sample B2 than B1 is due to the increased electric field resulted from the decreased gate-to-drain spacing by the inserted 2- μ m DFP structure, as shown in Fig. 1(c). Consequently, enhanced $g_{m,max}$ performances have been achieved by the improved current densities in the present samples B1 and B2. The devised MOS-HFETs are superior to our previous woks [15], [16] and other wide-gap-channel devices of $g_{m,max} = 16$ mS/mm and $I_{DS,max} = 90 \text{ mA/mm} [25], g_{m,max} = 9 \text{ mS/mm} \text{ and } I_{DS,max} =$ 350 mA/mm [29], $I_{DS,max} = 114$ mA/mm [30], $g_{m,max} =$ 38 mS/mm and $I_{DS,max} = 635$ mA/mm [31], $I_{DS,max} =$ 420 mA/mm [32], $g_{m,max} = 2.4$ mS/mm and $I_{DS,max} =$ 13 mA/mm [33], and $g_{m,max} = 97.9$ mS/mm and $I_{DS,max} =$ 473 mA/mm [34].

The gate-voltage swing (GVS) was defined to be the available V_{GS} range where the g_m value was within 90% of $g_{m.max}$. Sample A has wider GVS of 4.9 V than 2.9 (2.3) V of sample B1 (B2), indicating its good linearity. The threshold voltage (V_{th}) was defined as the V_{GS} intercept by the extrapolated line of $(I_{DS})^{1/2}$. Sample A1 has the lowest V_{th} of about -11 V as compared to -6.8 (-8.6) V of sample B1 (B2). The observed small V_{th} deviations from the C-V curves in Fig. 4 was due the different anode/cathode separation as compared to the MOS-HFETs. Higher GVS and lower V_{th} of sample A1 than sample B1 were due to its wider bandgap and higher conduction-band discontinuity (ΔE_C) of the In_{0.12}Al_{0.88}N barrier/buffer than those of In_{0.12}Al_{0.76}Ga_{0.12}N. Moreover, higher I_{DS} density of sample B2 has resulted in its lower V_{th} value than sample B1, since more negative V_{GS} was needed to deplete the channel. The specific on-resistances $(R_{on,sp})$ of samples A1, B1, and B2 were determined to be 5.14 m Ω /cm², 2.81 m Ω /cm², and 2.83 m Ω /cm². The present samples B2/B1 with/without DFP have shown comparable $R_{on,sp}$ values. They have also exhibited lower $R_{on,sp}$ than sample A1. Since $R_{on,sp}$ was extracted at low electric fields, reduced $R_{on,sp}$ was resulted from the improved channel conductivity by the devised In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/Al_{0.21}Ga_{0.79}N heterostructure. It is consistent with the characterized Hall data. The present devices has shown lower $R_{on,sp}$ than other AlGaN channel devices [25], [35]. Besides, the subthreshold swing (SS)



FIGURE 7. Two-terminal off-state BV_{GD} characteristics at 300 K for samples A1, B, and B2.



FIGURE 8. Three-terminal on-state BVDS characteristics at 300 K for samples A1, B, and B2.

and on/off-current ratios (I_{on}/I_{off}) for the studied samples A1, B1, and B2 were found to be 112.2 mV/dec and 2.2 \times 10^8 , 104 mV/dec and > 10^9 , and 124.3 mV/dec and > 10^9 , respectively. The studied devices have shown higher I_{on}/I_{off} ratios than other works [25], [29], [35]. It was due to the devised In_{0.12}Al_{0.88}N or In_{0.12}Al_{0.76}Ga_{0.12}N barrier/buffer, since I_{on} was increased by the improved self-polarization effect, whereas the I_{off} densities were reduced by the widegap barrier/buffer. Besides, improved gate insulation, $g_{m,max}$ gain, and I_{DS} density resulted from the MOS-gate design with high-k dielectric, wide-gap Al₂O₃, and good surface passivation have also contributed to the enhanced switching performance. Moreover, the I_{off} leakage in sample B2 was further suppressed by the DFP design. High Ion/Ioff and superior SS performances with decreased $R_{on,sp}$ of the present design are advantageous to the power-switching circuit applications.

The two-terminal off-state gate-drain breakdown voltage (BV_{GD}) and three-terminal on-state drain-source breakdown voltage (BV_{DS}) characteristics of samples A1, B1, and B2 at 300 K are shown in Figs. 7 and 8, respectively. BV_{GD} and BV_{DS} were determined to be the corresponding V_{GD} and V_{DS} biases where the I_{GD} and I_{DS} densities were equal to 1 μ A/mm. The source terminal remained float while characterizing BV_{GD} , and the V_{GS} was biased at -12 V for measuring BV_{DS} for all devices. BV_{GD} and BV_{DS} were found to be -410 V and 425 V, -465 V and 460 V, and -455 V and 515 V for samples A1, B1, and B2. Good breakdown characteristics have been

TABLE 1.	Device	characteristics	of t	he stu	died	samp	oles.
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Sample	A1	B1	B2	
IDSS0 (mA/mm)	628	755	791.1	
I _{DS, max} (mA/mm)	978.5	> 1000	> 1000	
$g_{m, max} (\mathrm{mS/mm})$	89.7	148.3	150.8	
V_{th} (V)	-11	-6.8	-8.6	
GVS (V)	4.9	2.9	2.3	
Ion/Ioff	2.2×10^{8}	> 109	> 10 ⁹	
SS (mV/dec)	112.2	104	124.3	
$R_{on, sp} (\mathrm{m}\Omega/\mathrm{cm}^2)$	5.14	2.81	2.83	
$BV_{GD}\left(\mathrm{V} ight)$	-410	-465	-455	
$BV_{DS}(V)$	425	460	515	
$\frac{BV_{DS}^2/R_{on, sp}}{(\text{mW/cm}^2)}$	35.1	75.4	93.7	

obtained in the studied devices. It is attributed by the reduced gate/substrate leakage currents by the devised MOS-gate structure with the USPD-grown Al₂O₃ surface passivation, wide-gap Al_{0.21}Ga_{0.79}N channel, and widegap barrier/buffer using In_{0.12}Al_{0.76}Ga_{0.12}N or In_{0.12}Al_{0.88}N compounds. About 13% improvement in BVGD was obtained in sample B1 with respect to sample A1. It is possibly due to the decreased gate leakage by improved interfacial quality of the devised In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/ Al_{0.21}Ga_{0.79}N heterostructure, as discussed before. Though BV_{GD} was slightly degraded in sample B2 due to reduced gate-to-drain separation, it has shown about 21% (12) improvement in BV_{DS} as compared to sample A1 (B1). It is because that the electric field distribution has been effectively smoothed out [21] by the DFP design. $BV_{DS}^2/R_{on,sp}$ is known as an important device figure-of-merit (FOM) for power switching applications. The present In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/Al_{0.21}Ga_{0.79}N MOS-HFET with (without) DFP design has shown superior device FOM of 93.7 (75.4) MW/cm² as compared to other AlGaN channel devices [25], [34], [35]. Table 1 summarizes the device characteristics of the studied devices. Superior current densities, high breakdown voltages, and improved switching characteristics have been successfully achieved in the present design.

IV. CONCLUSION

Wide-gap-channel In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/Al_{0.21}Ga_{0.79}N MOS-HFETs grown on a SiC substrate with/without DFP have been reported for the first time. Superior $I_{DS,max}$ in excess of 1 A/mm with low $R_{on,sp}$ has been achieved due to improved polarization effect, interfacial property, and carrier confinement by the In_{0.12}Al_{0.76}Ga_{0.12}N barrier/buffer design. The DFP was further integrated to improve the breakdown

performance. A reference In_{0.12}Al_{0.88}N/AlN/Al_{0.21}Ga_{0.79}N MOS-HFET was fabricated in comparison. The present In_{0.12}Al_{0.76}Ga_{0.12}N/AlN/ Al_{0.21}Ga_{0.79}N MOS-HFET design with (without) DFP has demonstrated superior I_{DSS0} of 791.1 (755) mA/mm, $I_{DS,max}$ of >1 (>1) A/mm, $g_{m,max}$ of 150.8 (148.3) mS/mm, GVS of 2.3 (2.9) V, I_{on}/I_{off} of >10⁹ (>10⁹), SS of 124.3 (104) mV/dec, $R_{on,sp}$ of 2.83 (2.81) m Ω/cm^2 , BV_{GD} of -455 (-465) V, BV_{DS} of 515 (400) V, and $BV_{DS}^2/R_{on,sp}$ of 93.7 (75.4) MW/cm². The present MOS-HFET design possessing high current drive capability, low R_{on} , and high device FOM is especially advantageous to shorten the battery charging time and improve the battery charging efficiency for the electric vehicle applications.

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