

2.7 kV Low Leakage Vertical PtO_x/ β -Ga₂O₃ Schottky Barrier Diodes With Self-Aligned Mesa Termination

Zhao Han^o[,](https://orcid.org/0009-0000-9602-3642) *Grad[u](https://orcid.org/0000-0001-9905-150X)at[e](https://orcid.org/0000-0002-2489-5617) Student Member, IEEE*, Guangzhong Jian, Xuanze Zhou^o, Qiming He^o, Weibin[g H](https://orcid.org/0000-0002-0456-6672)ao, *Graduate [Stu](https://orcid.org/0000-0001-5307-9187)dent Member, [IE](https://orcid.org/0000-0003-1611-3483)EE*, Jinyang Liu, Botong Li, Hong [Hu](https://orcid.org/0000-0001-6220-4461)ang, *Member, IEEE*, Qiuyan Liⁿ, Xiaolong Zhaoⁿ, Guangwei Xuⁿ, *Member, IEEE*, and Shibing Longⁿ, *Senior Member, IEEE*

*Abstract***— In this study, we fabricated superb** β**-Ga2O³ Schottky barrier diodes (SBDs) with high breakdown voltage (***V***br) and low leakage through combining platinum oxide (PtOx) and anodic self-aligned mesa termination (SAMT). The PtO^x that forms a high barrier with** β**-Ga2O³ enables the SAMT to function sufficiently. The in-situ annealing dry etch process repair the mesa sidewall and improve the Schottky contact well. SBDs with different mesa-etched depths (***D***ee) were systematically studied,** including 0, 0.3, 0.6, 0.9, and 1.2 μ m. The results showed **that the** V_{br} **of the PtO_X/** β **-Ga₂O₃ SBD increased from 1120 V to 2738 V, yielding a high power figure of merit (PFOM) of 1.02 GW/cm² . Meanwhile, the device maintained a less than 10** µ**A/cm² leakage current density until -2000 V. Devices with radii of 200, 100, and 50** µ**m obtained highest** V_{br} of 2508, 2772, and 2738 V at a D_{ee} of 1.2 μ m, respec**tively. The devices can be passivated by SU-8 without** *V***br degradation. This work provides an effective method for further improving the performance of** β**-Ga2O³ SBDs and promotes the application of** β**-Ga2O³ power diodes.**

*Index Terms***—** β**-Ga2O3, PtOx, breakdown voltage, power Schottky barrier diodes, self-aligned mesa termination.**

I. INTRODUCTION

T HE ultra-wide bandgap semiconductor $β$ -Ga₂O₃ is a promising material for manufacturing high power density promising material for manufacturing high power density and low loss electronic devices due to its wide bandgap and low loss electronic devices due to its wide bandgap $(\sim 4.5 \text{ eV})$, high critical breakdown field (8 MV/cm), and high Baliga's figure of merit (3400_{Si}) compared with GaN and SiC [\[1\],](#page-3-0) [\[2\],](#page-3-1) [\[3\]. Th](#page-3-2)e availability of melt-grown single crystal β -Ga₂O₃ substrates gives it attractive potential for low-cost and high-quality homoepitaxy [\[4\].](#page-3-3)

Manuscript received 10 July 2023; revised 7 August 2023; accepted 11 August 2023. Date of publication 15 August 2023; date of current version 27 September 2023. This work was supported in part by the NSFC under Grant 61925110, Grant U20A20207, Grant 62004184, Grant 62004186, and Grant 62234007; in part by the Key-Area Research and Development Program of Guangdong Province under Grant 2020B010174002; in part by the University of Science and Technology of China (USTC) Research Funds of the Double First-Class Initiative under Grant YD2100002009 and Grant YD2100002010; in part by the Fundamental Research Plan under Grant JCKY2020110B010; and in part by the Center for Micro and Nanoscale Research and Fabrication of USTC. The review of this letter was arranged by Editor E. Ahmadi. *(Zhao Han and Guangzhong Jian contributed equally to this work.) (Corresponding authors: Guangwei Xu; Shibing Long.)*

The authors are with the School of Microelectronics, University of Science and Technology of China (USTC), Hefei 230026, China (e-mail: xugw@ustc.edu.cn; shibinglong@ustc.edu.cn).

Color versions of one or more figures in this letter are available at https://doi.org/10.1109/LED.2023.3305389.

Digital Object Identifier 10.1109/LED.2023.3305389

Over the past decade, β -Ga₂O₃-based power electronic devices, especially vertical Schottky barrier diodes (SBDs), have developed rapidly. Most works aim at better electric field modulation, higher voltage/current application [\[5\], an](#page-3-4)d enhanced thermal management $[6]$. To pursue the theoretical limit of β -Ga₂O₃ SBD, various structures have been implemented to smooth the electric field distribution at the anode, including field plate $[7]$, $[8]$, mesa termination $[9]$, $[10]$, $[11]$, [\[12\], i](#page-3-11)mplanted edge termination [\[13\],](#page-3-12) [\[14\],](#page-3-13) [\[15\], t](#page-3-14)hermallyoxidized termination [\[16\],](#page-3-15) [\[17\], a](#page-3-16)nd junction barrier Schottky diodes [\[18\].](#page-3-17)

Among these, the mesa termination has an obvious performance improvement owing to avoiding lateral depletion region at anode edge. However, the leakage current of mesa-structured SBDs is of considerable relevance to the Schottky contact quality of active region. The leakage current through metal/Schottky contact would be hard to control due to the barrier lowering and tunneling effect under high electric field [\[19\]. M](#page-3-18)ulti-trench combined with metal-oxidesemiconductor structure has been confirmed to be a good solution for the leakage problem in the current p-type lack period [\[10\]. H](#page-3-9)owever, since most of the area of the trench SBDs is occupied by dielectric, the forward conduction characteristics are sacrificed. Meanwhile, the potential localized defects in the dielectric will involve the increased leakage current and reliability concerns such as time dependent dielectric breakdown (TDDB) under high electric fields [\[20\],](#page-3-19) [\[21\].](#page-3-20)

Increasing the barrier height of the Schottky contact is a simple way to reduce the leakage current. In particular, it will be more effective when the anode edge electric field is sufficiently weakened. Therefore, the combination of high Schottky barrier height and mesa termination is expected to achieve low leakage and high voltage SBD. In addition, the active area of the device would be utilized to the maximum. Recently, the metal-oxide electrode PtO_x has good application prospects in β -Ga₂O₃ power devices [\[22\],](#page-3-21) [\[23\]. T](#page-3-22)he barrier height of PtO_x/ β -Ga₂O₃ can reach more than 1.8 eV by controlling the oxygen flow during deposition [\[22\],](#page-3-21) [\[24\],](#page-3-23) [\[25\].](#page-3-24) It can be a good choice for the conceived device structure.

In this work, we achieved over 2.7 kV β -Ga₂O₃ SBDs based on the PtO_x and anodic self-aligned mesa termination (SAMT). Thanks to the high barrier height of the $PtO_x/$ β -Ga₂O₃ contact and the edge electric field modulation of the mesa structure, the leakage current density of the fabricated SBDs was suppressed effectively less than 10 μ A/cm² under -2000 V. The in-situ annealing dry etch process (I-AE)

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Fig. 1. (a) Cross-sectional schematic and fabrication details of β -Ga₂O₃ SBD with SAMT. (b) Carrier concentration extracted by $1/C² - V$ characteristics. (c) Cross-sectional SEM image at the anode edge of the 1.2 μ m-SBD.

reduced the surface charge on the sidewalls and significantly increased the breakdown voltage (V_{br}) . In addition, the relationship of electrical properties and electric field distribution to mesa-etched depths (D_{ee}) were systematically investigated.

II. DEVICE STRUCTURE AND FABRICATION

Fig. [1\(a\)](#page-1-0) schematically displays the device structure and fabrication details of the SBD featuring SAMT. The wafers used to fabricate the devices were taken from the same 2-inch epitaxial substrate with an epitaxial thickness about 10 μ m. The epitaxial structure was grown using halide vapor phase epitaxy (HVPE) on high-conductivity (001) substrates by Novel Crystal Technology, Inc., Japan. Fig. [1\(b\)](#page-1-0) shows the carrier concentration (N_d-N_a) extracted by capacitance-voltage (*C*-*V*) measurements of five devices at 100 *k*Hz before etching with an average doping about 1×10^{16} cm⁻³.

The fabrication process flow started from organic and acid cleaning. The Ti/Al/Ni/Au (40/200/50/50 nm) metal stacks were deposited onto the backside of the samples by electron beam evaporation (E-beam), followed by 470 ◦C rapid thermal annealing in N_2 for 1 minute. The PtO_x/Pt Schottky anode was patterned through photoresist and deposited in magnetron sputtering system. During P_1O_x sputtering, the power was set to 50 W, the chamber pressure was maintained at 5.2 mTorr. And the flow rates of O_2 and Ar were 20 and 40 sccm, respectively. The oxygen was turned off during sputtering of Pt. To reduce the resistance of PtO_x anode $[25]$, its thickness was rigorously determined. Finally, 12 nm-thick PtO_x and 84 nm-thick Pt were deposited sequentially at a rate of 1 Å/s.

The Ti/Ni (20/300 nm) metal stacks were deposited on Pt by E-beam with a rate of 1 Å/s as the hard mask for the selfaligned etch. Immediately, lift-off was performed to obtain patterned circular electrodes. Dry etching to form the mesa was performed in ICP 180 system with BCl₃/Ar gas flow of 35/5 sccm, 30 W RIE power, 900 W ICP power, and 5 mTorr chamber pressure. Pump oil (PO) was intentionally avoided to be applied between the Si carrier and β -Ga₂O₃ wafer so that the accumulated temperature (estimated about $150 °C$) on the wafer achieved both the post-annealing (PA) of Schottky contact and the in-situ annealing repair to the sidewall.

Four samples were etched at a rate of 100 nm/min for 3, 6, 9, and 12 minutes, obtaining different D_{ee} about 0.3, 0.6,

Fig. 2. *I*–*V* characteristics in (a) linear- and (b) log-scale of the SAMT-SBD with different etching depths. The V_{br} of the SAMT-SBDs (c) without passivation and (d) comparison before and after passivation. Inset in (b): extracted *n* and *q*Φ distributions with etching depth.

TABLE I PROPERTIES OF THE DEVICES

Device name	Ref- SBD	$0.3 \mu m$ SBD	$0.6 \ \mu m$ SBD	$0.9 \ \mu m$ SBD	$1.2 \ \mu m$ SBD
D_{ee} (μ m)	0	0.31	0.60	0.89	1.21
$R_{\text{on,sp}}$ (m Ω ·cm ²)	5.15	6.07	6.27	7.27	7.33
V_{on} (V)	1.67	1.50	1.48	1.43	1.45
$V_{\text{bi}}(V)$	1.751	1.503	1.484	1.491	1.564
$V_{\rm br}$ (V)	1120	1644	2128	2350	2738
$V_{\text{br-Ni}}$ (V)	654		1006	1304	
$PFOM$ (GW/cm ²)	0.24	0.45	0.72	0.76	1.02

All parameters listed in the table except $V_{\text{br-Ni}}$ pertain to the PtO_x devices.

0.9, and 1.2 μ m. The diodes are named as 0.3, 0.6, 0.9, and 1.2 μ m-SBD. Fig. [1\(c\)](#page-1-0) shows a scanning electron microscopy (SEM) cross-section of the 1.2 μ m-SBD sample, and the rounded corner profile exist at the bottom half of the etched sidewall.

Both forward current-voltage (*I-V*) and *C-V* measurements of the devices were performed by Keysight B1500A Semiconductor Device Analyzer. The *V*_{br} was measured using Keysight B1505A Power Device Analyzer with devices immersed in Fluorinert FC-770.

III. RESULTS AND DISCUSSION

Fig. [2\(a\)](#page-1-1) shows the linear plot of forward *I-V* characteristics and the extracted differential specific on-resistance $(R_{on,sp})$ of five D_{ee} of devices with a diode radius (*r*) of 50 μ m. For different D_{ee} , the $R_{on,sp}$, turn-on voltage (V_{on}) and build-in potential (*V*bi) are extracted, as shown in Table [I.](#page-1-2) The *V*on was extracted from extrapolation of the linear fits to $J = 0$ A/cm². The current density $(@5 V)$ decreases slightly with increasing *D*ee, which may be attributed to the decrease of the spreading current and depletion of etched-sidewall [\[26\].](#page-3-25) The inset in Fig. $2(b)$ shows the statistical results of the device ideality factor (n) and barrier height $(q\Phi)$ fitted by the thermionic emission (TE) model. The *n* and $q\Phi$ are 1.05, 1.88 eV respectively before etching. The decrease of $q\Phi$ after etching may be related to the variation of oxidation state of platinum atoms [\[27\],](#page-3-26) [\[28\]](#page-3-27) and the reduction of interface state [\[29\], b](#page-3-28)oth influenced by the temperature introduced during I-AE.

The reverse breakdown *I-V* characteristics for the SAMT-SBDs with different *D*_{ee} are shown in Fig. [2\(c\).](#page-1-1) The

Fig. 3. (a) Simulated electric field distributions of the SAMT-SBDs under 2000 V reverse bias. (b) Extracted electric field profile of the SBDs with different *D*_{ee} along the vertical (left) and lateral (right) directions respectively.

*V*br values are 1644, 2128, 2350, and 2738 V for the *D*ee of 0.3, 0.6, 0.9, and 1.2 μ m, respectively. Compared with Ref-SBD (without etching), the V_{br} is increased to 2.44 times. It is noteworthy that the device maintains a low leakage current density (below 10 μ A/cm²) until -2000 V, thanks to the high barrier of P_1O_x and the suppression of peak electric field by the SAMT. Table [I](#page-1-2) lists the device characteristics of the five types of SBDs. Besides, preliminarily results show that SAMT-SBDs can be passivated by SU-8 $\lceil 30 \rceil$ with almost unchanged V_{br} to meet practical applications (Fig. [2\(d\)\)](#page-1-1). Furthermore, Table [I](#page-1-2) also includes the *V*br of SAMT-SBDs using Ni anode, exhibiting significant gaps compared to the devices with PtO_x anode.

TCAD simulations in Silvaco Atlas are used to simulate the electric field distribution of the device at -2000 V, as shown in Fig. [3.](#page-2-0) According to the N_d - N_a values of Fig. [4\(a\),](#page-2-1) the surface charge is set to -3×10^{11} cm⁻² as estimated by assuming the depleted space-charge is balanced by the negative surface charge. As depicted in Fig. $3(b)$ -left, with increasing *D*ee, the peak electric field of the mesa sidewall decreases from 11.7 MV/cm to 5.5 MV/cm. Besides, the electric field distribution along Schottky interface becomes smoother (as shown in Fig. $3(b)$ -right). Nevertheless, the surface electric field approaching 3 MV/cm highlights the need for a stronger Schottky contact. Therefore, the high barrier formed by PtO_x/ β -Ga₂O₃ plays a crucial part in suppressing the leakage current and enhancing the *V*_{br} of the device.

Apart from the high Schottky barrier, another key factor for achieving high V_{br} is the repair of etched-sidewall to withstand the high electric fields existing there. The devices are labeled as Sample A, B, C, D, corresponding to the etching schemes of without etching, etching with PO, etching with PO & PA at 150 °C in N₂ for 12 minutes, etching without PO, respectively (Fig. $4(d)$). The $1/C^2$ -*V* curve of Sample B exhibits noticeable bending (Fig. [4\(a\)\)](#page-2-1). The extracted depletion depth and N_d - N_a at 0 V (649.7 nm, and 3.6 \times 10¹⁵ cm⁻³) quite differ from the values of Sample A (469 nm, and 8.9×10^{15} cm⁻³). It indicates a substantial surface charge located at the sidewall, which depletes the active region. Besides, for Sample A to B, the V_{br} of the devices show just 25% improvement (Fig. [4\(c\)\)](#page-2-1), speculating that the damage to the sidewall may be a primary factor limiting the device's performance.

For comparison, the Sample D with I-AE scheme can withstand V_{br} up to 2876 V, an increase of at least 180% compared to Sample A. And the bending of $1/C^2$ -V is apparently reduced (500.8 nm, and 6.4×10^{15} cm⁻³), almost consistent with Sample A. In addition, adding PA in Sample C improves the V_{br} while having little effect on the $1/\overline{C}^2$ -*V* curve. But the obvious *I-V* hysteresis reduction shown in Fig. [4\(b\)](#page-2-1) indicates the PA mainly affect the Schottky interface quality $[29]$. In conclusion, the significant improvement in V_{br}

Fig. 4. Comparison of (a) $1/C^2$ - *V* curves and extracted N_d - N_a and (b) hysteresis effect and (c) breakdown characteristics for devices with (d) different etching schemes.

Fig. 5. (a) Statistical distributions of V_{br} of SAMT-SBD with different anode radii (r) and etching depths. (b) $R_{on,sp}$ versus V_{br} benchmarks of reported state-of-the-art β -Ga₂O₃ SBDs.

of device with I-AE can be attributed to its dual contribution in sidewall charges and Schottky contact.

The statistics of V_{br} with different *r* are shown in Fig. [5\(a\),](#page-2-2) and further increasing D_{ee} has also been performed. The *V*br value shows an upward and then downward trend with increasing D_{ee} . The maximum V_{br} values obtained at 1.2 μ m with *r* value of 50 μ m, 100 μ m and 200 μ m are 2738 V, 2772 V and 2508 V, respectively, which are almost 2.4 times that of the corresponding size Ref-SBD. The trade-off point can be attributed to the accumulated damage introduced during etching, despite a monotonous increasing trend of the device's PFOM versus D_{ee} is shown based on our simulation.

The power figure of merit (PFOM) of SAMT-SBDs and Ref-SBD with r of 50 μ m is benchmarked against some stateof-the-art vertical and lateral SBDs in the plot of *R*on,sp versus *V*br in Fig. [5\(b\).](#page-2-2) Compared with Ref-SBD, the PFOM of 1.2 μ m-SBD is increased to 4.2 times (1.02 GW/cm²). In addition, the *V*br of devices with different sizes in this work is at the forefront of the reported work, demonstrating the potential of combining $P_{\text{t}}O_{\text{x}}$ and SAMT to achieve high performance.

IV. CONCLUSION

In summary, we have achieved high performance vertical β -Ga₂O₃ SBD through anodic self-aligned mesa etching and the meaningful adoption of P_tO_x . The high Schottky barrier induced by PtO_x endows the SAMT fully realize its function. And the I-AE process combines the dual effects of sidewall repairing and post-annealing. The device maintains a low leakage current density $(<10 \mu A/cm^2)$ at -2000 V and achieves high PFOM up to 1.02 GW/cm². This work provides a practical and effective solution for enhancing the performance of etched terminal devices.

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