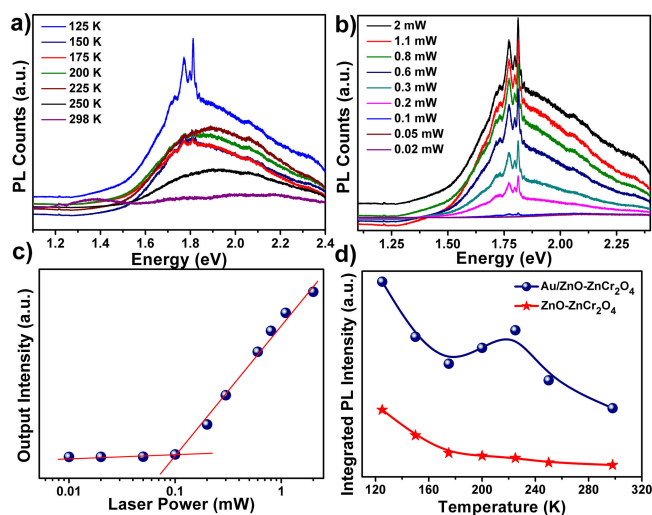


Exciton Lasing in ZnO-ZnCr₂O₄ Nanowalls

Volume 11, Number 6, December 2019

Tejendra Dixit
Jitesh Agrawal
Miryala Muralidhar
Masato Murakami
Kolla Lakshmi Ganapathi
Vipul Singh
M. S. Ramachandra Rao



DOI: 10.1109/JPHOT.2019.2945010

Exciton Lasing in ZnO-ZnCr₂O₄ Nanowalls

Tejendra Dixit ^{1,2}, Jitesh Agrawal ³, Miryala Muralidhar ⁴,
Masato Murakami,⁴ Kolla Lakshmi Ganapathi ¹, Vipul Singh ³,
and M. S. Ramachandra Rao ^{1,2}

¹Materials Science Research Centre and Department of Physics, Indian Institute of Technology, Madras, Chennai 600036, India

²Nano Functional Materials Technology Centre, Indian Institute of Technology, Madras, Chennai 600036, India

³Department of Electrical Engineering, Indian Institute of Technology, Indore, Indore 453552, India

⁴Superconducting Material Laboratory, Graduate School of Science and Engineering, Shibaura Institute of Technology, Tokyo 135-8546, Japan

DOI:10.1109/JPHOT.2019.2945010

This work is licensed under a Creative Commons Attribution 4.0 License. For more information, see <https://creativecommons.org/licenses/by/4.0/>

Manuscript received August 11, 2019; revised September 22, 2019; accepted September 28, 2019. Date of publication October 1, 2019; date of current version November 7, 2019. The paper was supported in part by Japan Student Services Organization (JASSO) for the Advanced Project Based Learning (aPBL), in part by Shibaura Institute of Technology (SIT) under the Top Global University Project, designed by the Ministry of Education, Culture, Sports, and Science & Technology in Japan. The work of T. Dixit was supported by the Indian Institute of Technology Madras through Institute Postdoctoral Fellowship. The work of K. L. Ganapathi was supported by the Department of Science and Technology, India, through sanction order no. DST/INSPIRE/04/2016/001865 under the DST INSPIRE Faculty program. Corresponding author: M. S. Ramachandra Rao (e-mail: msrrao@iitm.ac.in).

Abstract: We demonstrate low power continuous wave, red and NIR exciton lasing with FWHM of 1 nm, quality factor of 680 and threshold power of 100 μ W in ZnO-ZnCr₂O₄ nanowalls. The NIR lasing was enabled by integrating ZnO with ZnCr₂O₄. Moreover, wavelength selective photoluminescence (tuning from UV to NIR) and enhanced two-photon emission were also observed in ZnO-ZnCr₂O₄ nanowalls. The exciton-exciton scattering can be attributed to the observation of exciton lasing at low temperature (<200 K). A plausible mechanism has been elucidated in order to explain the results. This work will open new opportunities in the advancement of oxide semiconductors based exciton lasers.

Index Terms: Photoluminescence spectroscopy, exciton-plasmon coupling, direct band transition, multi-layer MoS₂.

1. Introduction

There are two major techniques of lasing i.e., conventional photon lasing and exciton and/or polariton lasing. While, the threshold power is restricted by the population inversion of the electron-hole plasma in the earlier, the scattering between excitons play significant role for the generation of exciton lasing. In particular, pure exciton lasing has received substantial consideration in recent years due to various reasons like sub-Mott density, low-threshold operation, etc.

ZnO is one of the most attractive semiconductor for the exploration of exciton lasing among scientific community due to several factors like ease of fabrication, excellent optoelectronic properties like high exciton binding energy (60 meV) etc. In the sequence to get room temperature exciton lasing, ZnO microcrystals have been employed in recent years. R. Matsuzaki *et al.* have demon-

strated that pure exciton lasing i.e., lasing owing to exciton-exciton scattering is the dominating lasing mechanism from very low temperature to 150 K [1]. Moreover, they found that above 150 K the lasing process will be governed by exciton-electron or exciton-LO phonon scatterings. Y. Y. Lai *et al.* have also found that the exciton lasing is stable below 140 K in ZnO microcavity [2]. As far as the ZnO nanostructures are concerned realization of lasing due to exciton-exciton scattering is difficult to achieve at room temperature, which has been demonstrated by C. Klingshirm *et al.* [3], [4]. Here it must be noted that the exciton lasing has been explored for ZnO samples in the UV spectral region, only.

As ZnO is a potential contender for optoelectronic industry therefore, its applicability for visible and near infrared (NIR) region mainly for exciton lasing requires attention in order to compete with existing Si based technology [5], [6]. In sequence to get red and/or NIR emission in ZnO the typical technique is to dope a rare earth metal which makes the process complex and costly [7]. It is well known that Cr³⁺ ions are responsible for the red and NIR emission and have been used in various systems [8]. Although, Cr doping can systematically tune the emission properties of ZnO, however, hitherto there have been no reports on the realization of exciton lasing in Cr incorporated ZnO systems as has been observed in the present study. For the incorporation of Cr we have used our previous work in which we have observed that the integration of ZnO with ZnCr₂O₄ brings new functionality. Interestingly, neither ZnO nor ZnCr₂O₄ have shown any signature of red and NIR lasing, individually.

In this work, we demonstrate red and NIR exciton lasing in Au coated ZnO-ZnCr₂O₄ nanowalls. Systematic temperature dependent photoluminescence study has verified the involvement of exciton-exciton scattering at low temperatures (below 200 K). Here, it is important to note that the nanowalls configuration is required for achieving low threshold lasing. The formation of nanowalls in our case has helped to achieve low threshold lasing power density. Along with the NIR emission and low threshold operation another interesting feature is continuous wave (CW) mode operation. All the previous studies have used pump optical excitation to achieve exciton lasing in ZnO system. It is important to note that CW excitation has several advantages like low cost operation, better output stability etc. and therefore more focus on exciton lasing using CW excitation is needed [9]. This work will open new avenues in the field of pure exciton lasing.

2. Experimental Details

ZnO-ZnCr₂O₄ nanowalls were grown through solution process [10]. At first, ZnO seed layer (thin film) coating was done by spin coating a colloidal (stable) solution of zinc acetate, ethanolamine in 2-methoxyethanol on the pre-cleaned glass substrate. After that, seed layer coated samples were annealed at 250 °C in air for 5 minutes. Towards the nanowalls growth, the seed layer coated samples were kept at the bottom of the beaker filled with a solution prepared by mixing 0.1 M Zn(NO₃)₂ · 6H₂O with equi-molar hexamethylenetetramine ((CH₂)₆N₄, also known as HMTA) in de-ionized (DI) water (25 mL). For the growth of ZnO-ZnCr₂O₄ nanowalls, 5 mM K₂Cr₂O₇ was added to the precursor solution. Finally, the beaker was sealed tightly and kept in an oven at 110 °C for 3 h for the uniform growth of the nanostructures. Finally, the samples were thoroughly washed with DI water to remove impurities [10]. For the analysis of emission spectra, the photoluminescence (PL) measurements were carried out using a continuous wave light source (λ value of 275 nm, 305 nm, 400 nm and 488 nm) from HORIBA LabRAM. In case of 488 nm laser; the laser spot size was nearly 1 μ m. Time resolved PL measurements were done using Delta-Pro TCSPC Lifetime Fluorometer (HORIBA). Perkin Elmer (LAMBDA 950) spectrophotometer was used to record the UV-visible absorbance spectra (from 200 to 2500 nm range).

3. Results

Scanning electron microscope images of ZnO-ZnCr₂O₄ nanostructured film (recorded using FESEM-Zeiss Supra-55) is shown in Fig. 1(a). Formation of oriented nanowalls (average thickness of nanowalls were ~35 nm) can be clearly seen. The mechanism of nanowall growth is

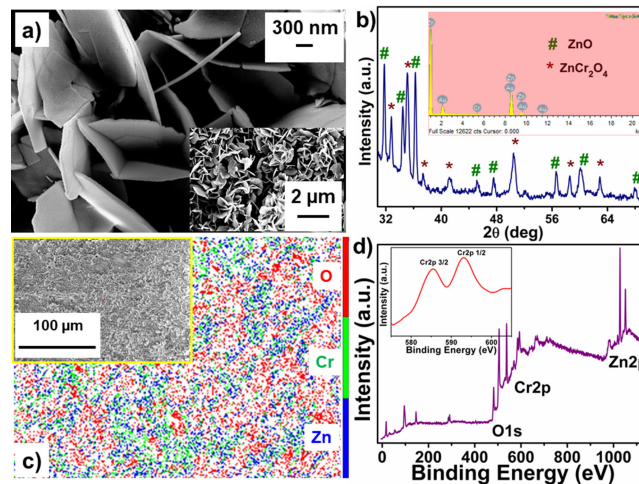


Fig. 1. (a) Scanning electron microscope image of ZnO-ZnCr₂O₄ nanowalls, inset shows the film morphology at low magnification; (b) X-ray diffraction patterns of the nanowalls, inset shows EDX spectra; (c) EPMA image (red, green and blue colors corresponds to Oxygen, Chromium and Zinc atoms, respectively); (d) High resolution X-ray photoelectron spectra.

mentioned in our previous work [10]. Concisely, addition of potassium dichromate plays a major role in the formation of nanowalls rather than typical nanorod structures. It was found that the nanowalls are broader at substrate and thin at the top i.e., making tapered nanowalls. The inset shows the morphology at low magnification. Powder-XRD (Rigaku Smart Lab system) plot for the nanowalls is shown in Fig. 1(b). Both, ZnO [ICDD#89–0510] and ZnCr₂O₄ [ICDD#22–1107] phases were clearly observed [11]. The electron dispersive spectrum (EDS) of Au/ZnO-ZnCr₂O₄ nanowalls is shown in the inset of Fig. 1(b), where all the elements can be clearly seen. For further investigating the elemental distribution, Electron Probe Micro Analyzer (EPMA) analysis was carried out using EPMA-JEOL: JXA-8530F system. EPMA results have shown that the elements were well distributed and thus it can be concluded that Cr is uniformly distributed in the matrix. The presence of Cr³⁺ ions in the matrix has been confirmed by high resolution XPS spectrum, which is shown in Fig. 1(d).

Fig. 2(a) shows the photoluminescence spectra of ZnO-ZnCr₂O₄ samples excited with three different wavelengths i.e., 275 nm, 305 nm and 400 nm. The PL spectra show a systematic modulation in the emission with the variation of excitation wavelength. In particular, with 275 nm excitation we were able to excite ZnCr₂O₄ phase with emission spectra encompassing UV, blue and green region of the electromagnetic spectra. The peak in the UV region i.e., between 320 nm-350 nm can be assigned to the direct band transitions of ZnCr₂O₄ [10]. With 305 nm excitation we were able to get broad emission covering UV, blue, green and orange spectral ranges. Interestingly, with 400 nm excitation we have observed strong blue-green and a signature of red emission. Visible region emissions are related to various defect related states of ZnO viz. oxygen vacancies (V_o), oxygen interstitials (O_i) etc. The visible region emissions can also be consigned to the several transitions associated to Cr³⁺ ions. Due to crystal field, the d-orbitals in the Cr³⁺ ions get split into *t*₂ and *e*-levels, thereby causing various transitions in the UV and visible region [12]. The similar trend has been reported in Si based devices and our results show that the ZnO-ZnCr₂O₄ nanowalls can also be used for visible and NIR emission [13], [14].

Interestingly, the sub-bandgap excitation (here, 488 nm laser has been used) shows amazing features i.e., NIR emission with peak position at 1.35 eV and 1.7 eV. Surprisingly, after Au coating we have found a significant enhancement in the NIR emission. The emission at 1.7 eV can be assigned to the typical 2E to ⁴A₂ transitions in the Cr³⁺ ions. The 1.35 eV transition can be assigned to the band formation and low energy excitons associated with the 2E states. The time resolved PL spectra is shown in Fig. 2(c). Generally rare earth doped ZnO shows life time of μs

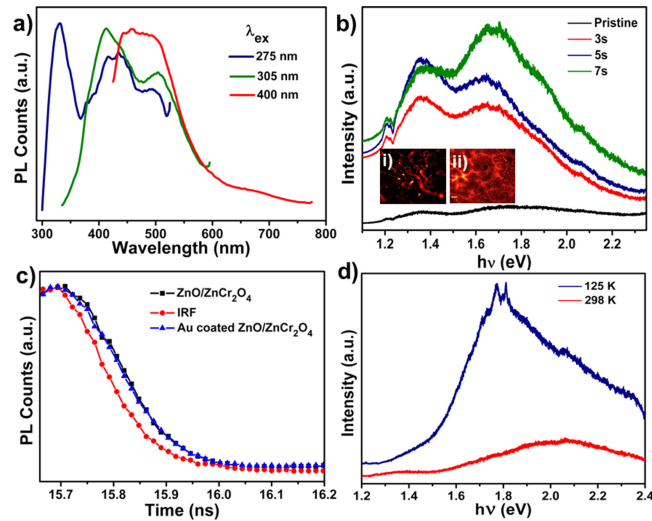


Fig. 2. (a) Room temperature PL spectra of ZnO-ZnCr₂O₄ nanowalls with various excitations; (b) PL spectra of Au coated ZnO-ZnCr₂O₄ nanowalls for different Au sputtering time (excited with 488 nm laser source), the inset show two-photon emission image (excited with 800 nm) of (i) ZnO-ZnCr₂O₄ nanowalls, (ii) Au/ZnO-ZnCr₂O₄ nanowalls; (c) Time resolved PL spectra; (d) PL spectra of pristine ZnO-ZnCr₂O₄ nanowalls at 298 K (red curve) and 125 K (blue curve).

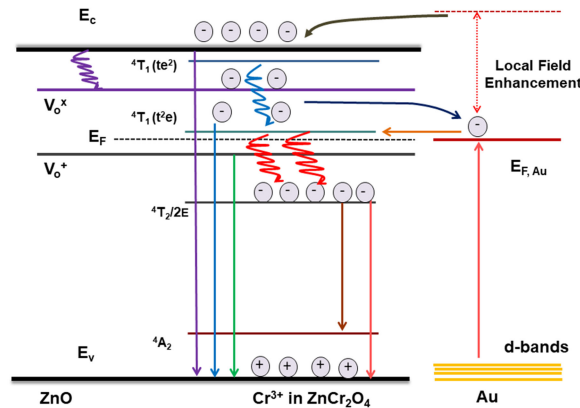


Fig. 3. Schematic for the charge transfer due to Au incorporation and energy levels responsible for emission.

while in our case it is in the range of several ps, which clearly suggests the involvement of excitonic transitions [1]. It must be noted here that, wiggles were perceived in the PL spectra for the Au coated samples. The intensity of PL was observed to increase with Au sputtering time and found to start decrease with 10 seconds of sputtering. The observation of wiggles in the spectra motivated us to investigate more. In that regard, we have taken PL spectra of the pristine samples at low temperatures. A clear transformation in the PL line shape can be seen at 125 K with two distinct peaks (as shown in Fig. 2(d)). This corresponds to the red and NIR transition due to the typical 2E to ⁴A₂ transitions in the Cr³⁺ ions. Pristine and Au/ZnO-ZnCr₂O₄ nanowalls have shown two-photon emissions also (excited with two-photons of 800 nm light source). Interestingly with Au coating the emission efficiency turned out to be more prominent in comparison to the pristine samples (shown in the inset of Fig. 2(b)). The charge transfer from Au to Cr³⁺ energy levels can be assigned to enhancement in NIR emission. The schematic for the energy levels of Cr³⁺ ions and interaction with Au is shown in Fig. 3.

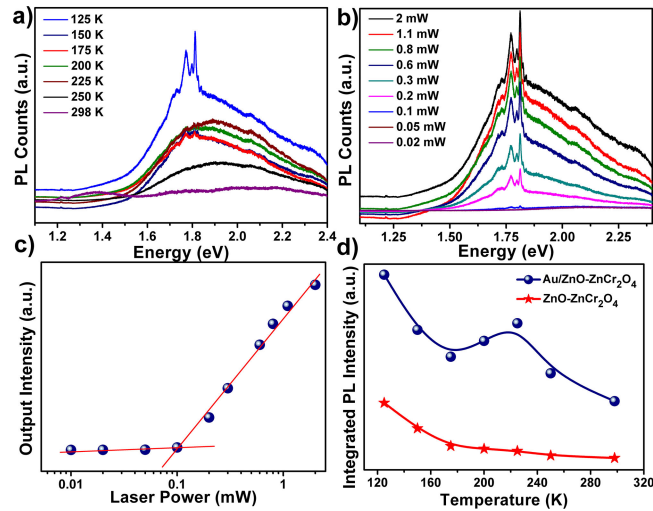


Fig. 4. Temperature dependent PL spectra of Au/ZnO-ZnCr₂O₄ nanowalls; (b) Excitation power dependent PL spectra of Au/ZnO-ZnCr₂O₄ nanowalls recorded at 125 K; (c) PL intensity vs. excitation power of Au/ZnO-ZnCr₂O₄ nanowalls recorded at 125 K. The lasing threshold has been estimated to be $\sim 100 \mu\text{W}$; (d) Integrated PL intensity versus temperature.

Temperature dependent PL spectra of Au/ZnO-ZnCr₂O₄ samples show several interesting features (shown in Fig. 4(a)). We can see that the two distinct peaks at 1.77 and 1.81 eV at 200 K and 150 K, which become dominant at 125 K with an additional sharp peak at 1.79 eV. The full width at half maximum (FWHM) for the peak in the red region was 1.3 nm and in the NIR region was 1.8 nm. The additional peak at 1.79 eV is however missing in pristine ZnO-ZnCr₂O₄ samples at 125 K and can be assigned to the interaction of Au with Cr³⁺ transitions and due to exciton-exciton scattering. The excitation intensity dependent PL spectra at 125 K were recorded and are shown in Fig. 4(b). Fig. 4(c) presents the excitation power dependent emission in the red and NIR region by varying laser power in the range 0.01 mW to 2 mW. It was found that the emission intensity increased abruptly beyond excitation power of 0.2 mW along with the emergence of sharp spike-like peaks. At lower excitation power (< 0.2 mW), no such sharp peaks were observed. Such a low lasing threshold can be assigned to (a) exciton-exciton scattering process and (b) Nanowall morphology. Nanowalls with tapered thickness have been demonstrated by E. S. Jang *et al.* to produce lasing with very threshold power density [15]. Oriented nanowalls can serve as natural waveguide and thus invoke the lasing at low excitation powers. The improvement of internal reflection mostly at the top surface is helping to reach lasing with ease due to Au coating.

The lasing can be assigned either to excitonic process or electron-hole plasma (EHP) process, generally. It is well known that the EHP state is formed once the density of electron-hole pairs exceeds the critical value of Mott density defined as n_M . There is no shift observed in the peak position of any of the emissions with excitation power even at $\sim 20 I_{ex}^{th}$, which clearly discard the possibility of EHP process as a reason or the lasing [3]. From the temperature dependent PL spectra it is clear that lasing is happening below a certain temperature which is possible only due to exciton-exciton scattering. In case of exciton-exciton scattering typically, one exciton goes through radiative recombination, whereas the other exciton is scattered to higher excitonic states i.e., $n = 2, 3, 4, \dots, \infty$. Additionally, this exciton-exciton scattering induces stimulated emission. Moreover, the wall length (few microns) is more than the wavelength of emitted light, which means the structure can support the lasing. The transition from exciton-exciton scattering to exciton-electron scattering has been assigned for the observation of lasing action in ZnO microcrystals by Matsuzaki *et al.* with the variation of temperature [1]. They demonstrated that the value of $E_{ex}(T) - \hbar\omega_{max}^{obs}(T)$ follows different trends with temperature for exciton-exciton and exciton-electron scattering processes. Where, $E_{ex}(T)$ is the free exciton transition energy at a given temperature and $\hbar\omega_{max}^{obs}(T)$ is the peak

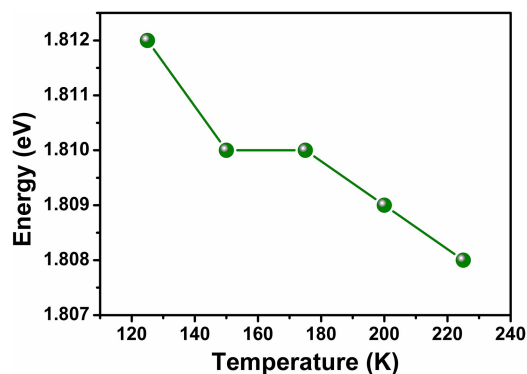


Fig. 5. Variation of $E_{ex}(T) - \hbar\omega_{max}^{obs}(T)$ with temperature.

energy as a function of temperature. In our case we have not observed any significant variation in the peak positions with temperature (the change in the peak present at 1.81 eV is nearly 4 meV even at 225 K to that of 125 K), which is shown in Fig. 5. These observations clearly suggest the presence of exciton-exciton scattering even at 225 K in Au/ZnO-ZnCr₂O₄ samples [1]. This system provides stable exciton-exciton scattering even at 200 K that can provide lasing.

The temperature-dependent integrated PL intensities of the pristine and Au coated ZnO-ZnCr₂O₄ nanowalls is shown in Fig. 4 (d). For pristine samples the plot follows Arrhenius formula, while for Au coated samples have shown a knee feature i.e., the intensity gradually increases as the temperature increases for a range of temperatures, which cannot be explained with Arrhenius formula [16], [17]. Similar trends have been shown by T. Lu *et al.* in case of GaN-based light-emitting diodes [17]. They have observed that the anomalous optical phenomenon is associated with the carrier transport behavior referred to as carrier mobility. Moreover, in case of optical excitation only, the observation of such an anomalous behavior is interesting. The temperature dependent charge transfer from Au to Cr³⁺ energy states and also the temperature dependent energy transfer from ZnO phase could play a role in this anomalous behavior. However, the origin of which is still under investigation.

4. Conclusions

Pure exciton lasing in red and NIR range has been shown in ZnO based system. Au coated ZnO-ZnCr₂O₄ nanowalls have been explored for various applications like red and NIR lasing and enhanced two-photon emission. The Au coated ZnO-ZnCr₂O₄ nanowalls have shown interesting feature of red and NIR lasing for the first time, which has been attributed to the exciton-exciton scattering at low temperature. We found that the exciton-exciton scattering is the leading effect for the lasing even at 225 K. ZnO-ZnCr₂O₄ system can be a potential candidate for exciton lasing and can further prompt the development of this field.

Acknowledgement

The authors would like to thank SAIF, IIT Madras for the usage of UV-Visible-NIR absorption and TRPL facility. JA is thankful to UGC, India for providing the fellowship (JRF). T. Dixit, K. L. Ganapathi, and M. S. Ramachandra Rao would like to thank Department of Science and Technology (DST) that led to the establishment of Nano Functional Materials Technology Centre (SR/NM/NAT/02–2005).

References

- [1] R. Matsuzaki *et al.*, "Purely excitonic lasing in ZnO microcrystals: Temperature-induced transition between exciton-exciton and exciton-electron scattering" *Phys. Rev. B*, vol. 96, no. 12, Sep. 2017, Art. no. 125306.

- [2] Y. Y. Lai, Y. H. Chou, Y. P. Lan, T. C. Lu, S. C. Wang, and Y. Yamamoto, "Crossover from polariton lasing to exciton lasing in a strongly coupled ZnO microcavity," *Sci. Rep.*, vol. 6, pp. 1–7, Feb. 2016.
- [3] C. Klingshirn, R. Hauschild, J. Fallert, and H. Kalt, "Room-temperature stimulated emission of ZnO: Alternatives to excitonic lasing," *Phys. Rev. B*, vol. 75, no. 11, Mar. 2007, Art. no. 115203.
- [4] M. A. M. Versteegh, D. Vanmaekelbergh, and J. I. Dijkhuis, "Room-Temperature laser emission of ZnO nanowires explained by many-body theory" *Phys. Rev. Lett.*, vol. 108, no. 15, Apr. 2012, Art. no. 157402.
- [5] M. J. Chen *et al.*, "Stimulated emission in a nanostructured silicon pn junction diode using current injection," *Appl. Phys. Lett.*, vol. 84, no. 12, pp. 2163–2165, Mar. 2004.
- [6] K. Xu *et al.*, "Light emission from a poly-silicon device with carrier injection engineering," *Materials Sci. Eng. B*, vol. 231, pp. 28–31, Apr. 2019.
- [7] R. Khanum, R. S. Moirangthem, and N. M. Das, "Observation of defect-assisted enhanced visible whispering gallery modes in ytterbium-doped ZnO microsphere," *J. Appl. Phys.*, vol. 121, no. 2, May 2017, Art. no. 213101.
- [8] M. Garcia-Teceador, D. Maestre, A. Cremades, and J. Piqueras, "Growth and characterization of Cr doped SnO₂ microtubes with resonant cavity modes," *J. Mater. Chem. C*, vol. 4, no. 24, pp. 5709–5716, May 2016.
- [9] S. Biswas and P. Kumbhakar, "Continuous wave random lasing in naturally occurring biocompatible pigments and reduction of lasing threshold using triangular silver nanostructures as scattering media," *Nanoscale*, vol. 9, no. 47, pp. 18812–18818, Nov. 2017.
- [10] T. Dixit, I. A. Palani, and V. Singh, "Investigation on the influence of dichromate ion on the ZnO nano-dumbbells and ZnCr₂O₄ nano-walls," *J. Mater. Sci. Mater. Electron.*, vol. 26, no. 2, pp. 821–829, Nov. 2014.
- [11] T. Dixit, J. Agrawal, K. L. Ganapathi, V. Singh, and M. S. R. Rao, "High-performance broadband photo-detection in solution-processed ZnO-ZnCr₂O₄ nanowalls," *IEEE Electron Device Lett.*, vol. 40, no. 7, pp. 1143–1146, Jul. 2019.
- [12] T. Dixit, A. Kumar, I. A. Palani, and V. Singh, "Surface-plasmon-mediated red and near infrared emission from Au-coated ZnO/ZnCr₂O₄ nanocomposites," *Scr. Mater.*, vol. 114, pp. 84–87, Mar. 2016.
- [13] C. H. Hsiao *et al.*, "Emitting layer thickness dependence of color stability in phosphorescent organic light-emitting devices," *Organic Electron.*, vol. 11, pp. 1500–1506, Jul. 2010.
- [14] Kaikai Xu, "Silicon MOS optoelectronic micro-nano structure based on reverse-biased PN junction," *Physica status solidi a*, vol. 216, no. 7, Apr. 2019, Art. no. 1800868.
- [15] E. S. Jang *et al.*, "Soft-solution route to ZnO nanowall array with low threshold power density," *Appl. Phys. Lett.* vol. 97, no. 4, Jul. 2010, Art. no. 043109.
- [16] Z. F. Shi *et al.*, "Near-infrared random lasing realized in a perovskite CH₃NH₃PbI₃ thin film," *J. Mater. Chem. C*, vol. 4, no. 36, pp. 8373–8379, Aug. 2016.
- [17] Z. Ma *et al.*, "Temperature-dependent photoluminescence in light-emitting diodes," *Sci. Rep.*, vol. 4, no. 6131, pp. 1–7, Aug. 2014.